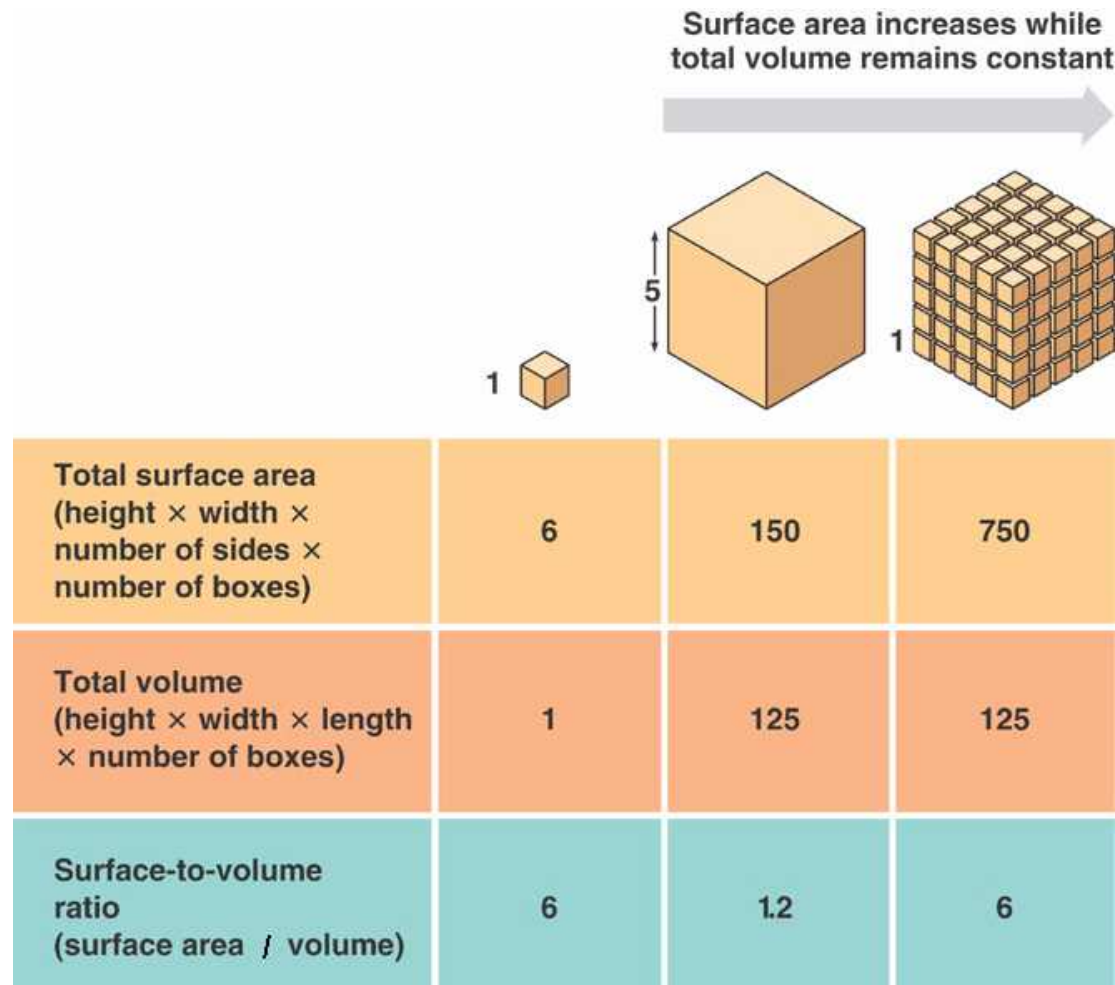


# Nanomaterials

- Metals and Alloys
  - Fe, Al, Au
- Semiconductors
  - Band gap, CdS, TiO<sub>2</sub>, ZnO
- Ceramic
  - Al<sub>2</sub>O<sub>3</sub>, Si<sub>3</sub>N<sub>4</sub>, MgO, , SiO<sub>2</sub>, ZrO<sub>2</sub>
- Carbon based
  - Diamond, graphite, nanotube, C60, graphene
- Polymers
  - Soft mater, block co-polymer
- Biological
  - Photonic, hydrophobic, adhesive,
- Composites

# Surface to Volume Ratio



# Surface Energy

One face surface energy:  $\gamma$

27 cube:  $27 \times 6 \gamma$

3 x 9 cube line:  $114 \gamma$

3 x (3x3) square:  $90 \gamma$

3 x 3 x 3 cube:  $54 \gamma$

# Surface to Volume Ratio

Au: AAA

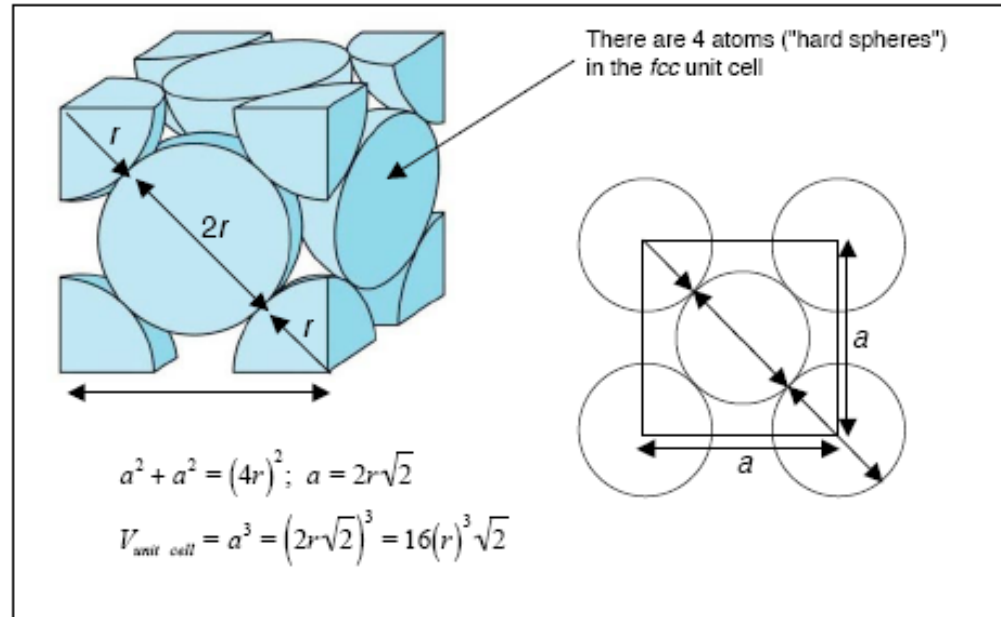
Atomic mass: 196.967

Density 19.31

Radii =0.144 nm

Number of Au atoms in 1 m	$3.4 \cdot 10^9$
Volume of Au atom	$4.19 \cdot 10^{28}$
Surface area Au atom	$7.22 \cdot 10^{19}$
Surface/volume ratio	$1.72 \cdot 10^{-9}$

# fcc



$$V_{\text{unit cell}} = a^3 = (2r\sqrt{2})^3 = 16(0.5\text{nm})^3\sqrt{2} = \mathbf{2.828 \text{ nm}^3}$$

$$\frac{10^{27} \text{ nm}^3}{2.828 \text{ nm}^3} = \mathbf{3.536 \times 10^{26} \text{ nano unit cells}}$$

$$\frac{S_{\text{spheres}}}{S_{\text{unit cell}}} = \frac{4.44 \times 10^9 \text{ m}^2}{6.0 \times 10^9 \text{ m}^2} = \mathbf{0.74}$$

$$\text{Collective Area} = 3.536 \times 10^{26} \text{ nano unit cells} \left( \frac{4 \text{ spheres}}{\text{unit cell}} \right) \left( \frac{4\pi r^2}{\text{sphere}} \right) = \mathbf{4.44 \times 10^{27} \text{ nm}^2}$$

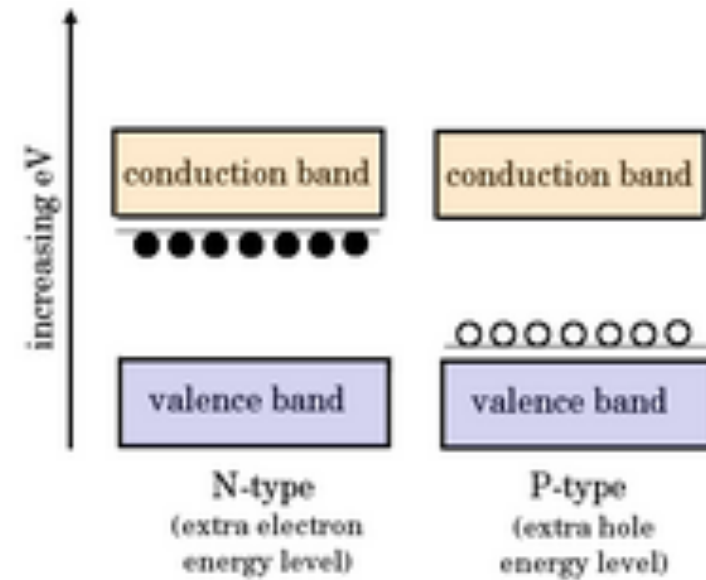
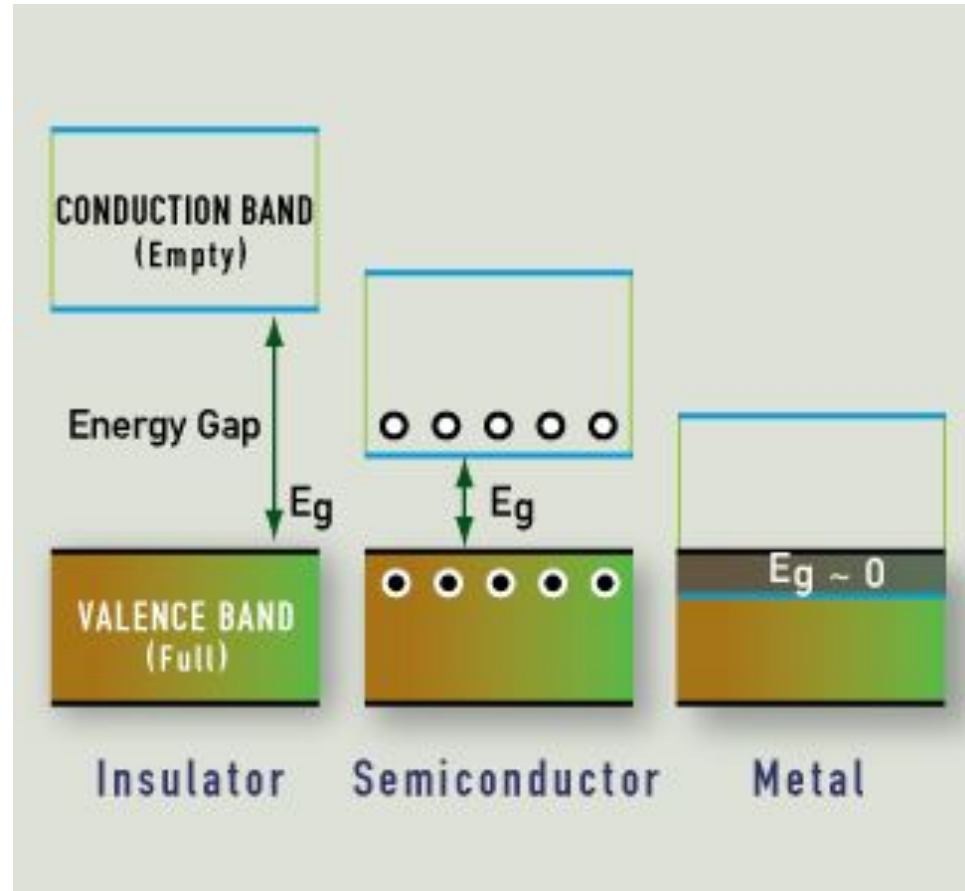
# Packing Fraction

$$\text{APF} = \frac{N_{\text{atoms}} V_{\text{atom}}}{V_{\text{crystal}}}$$

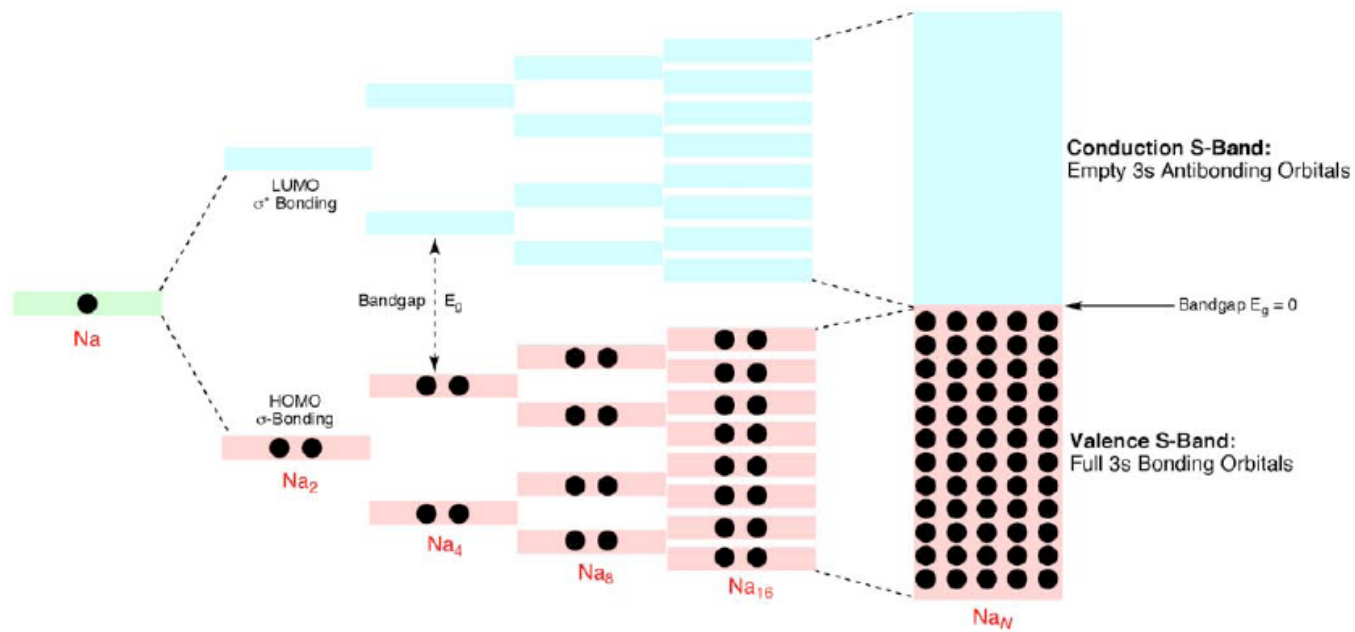
# Surfaces

- Collective surface area of nanocube 1 nm
- Porous materials
  - Micropore (<2 nm)
  - Mesopore (2 nm ~ 50 nm)
  - Marcopore (> 50nm)
- Void volume
  - $V_{\text{pore}}/V_{\text{material}}$

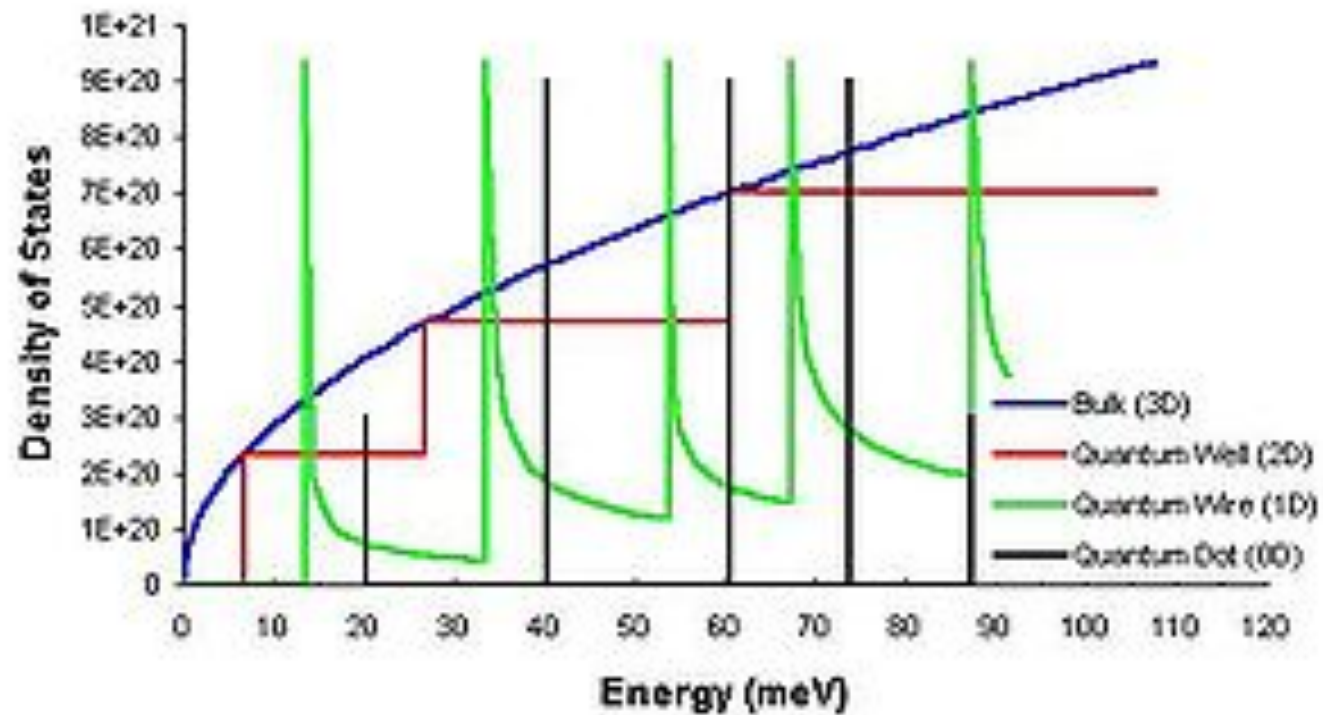
# Bandgap



# Bandgap



# Density of State

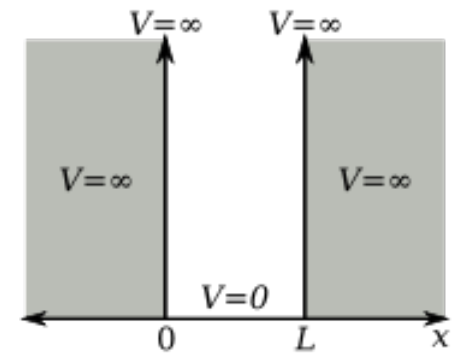


# Particle in a Box

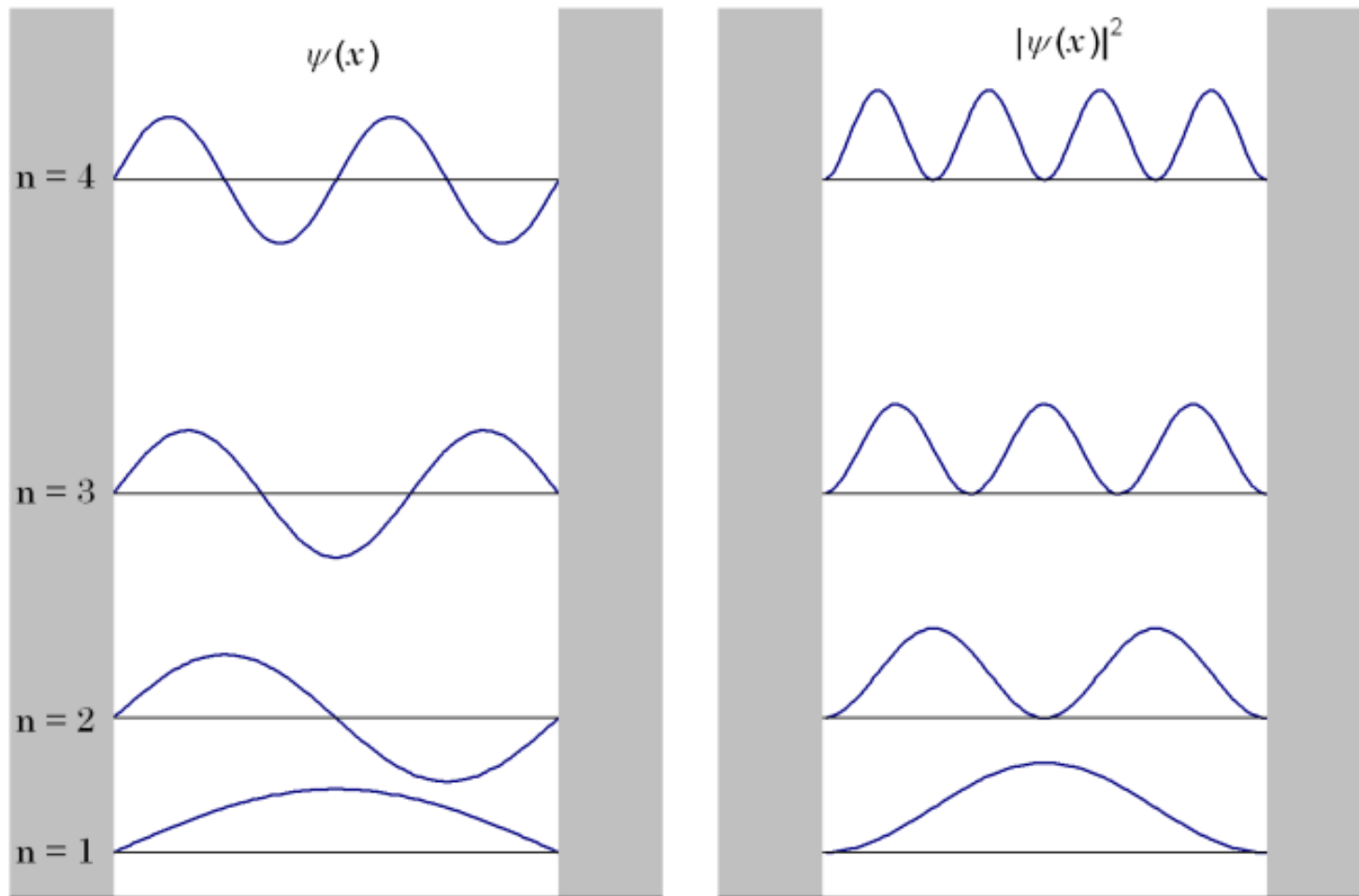
$$-\frac{\hbar^2}{2m} \frac{d^2\psi(x)}{dx^2} + V(x)\psi(x) = E\psi(x) \quad (1)$$

$$\psi_n = \sqrt{\frac{2}{L}} \sin\left(\frac{n\pi x}{L}\right)$$

$$E_n = \frac{\hbar^2 \pi^2}{2mL^2} n^2$$



# Particle in a Box



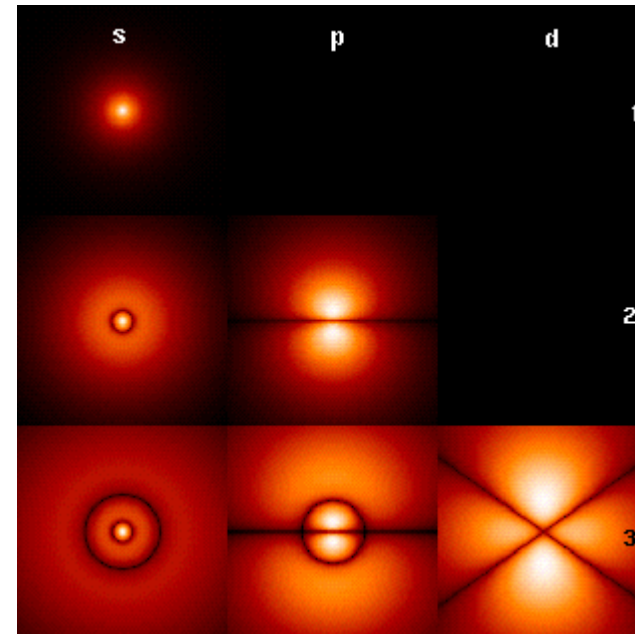
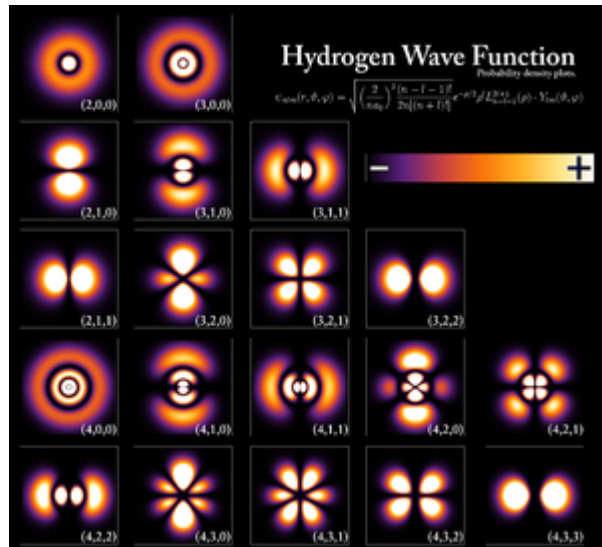
$$\psi_{n_x, n_y} = \sqrt{\frac{4}{L_x L_y}} \sin\left(\frac{n_x \pi x}{L_x}\right) \sin\left(\frac{n_y \pi y}{L_y}\right)$$

$$E_{n_x, n_y} = \frac{\hbar^2 \pi^2}{2m} \left[ \left(\frac{n_x}{L_x}\right)^2 + \left(\frac{n_y}{L_y}\right)^2 \right]$$

$$\psi_{n_x, n_y, n_z} = \sqrt{\frac{8}{L_x L_y L_z}} \sin\left(\frac{n_x \pi x}{L_x}\right) \sin\left(\frac{n_y \pi y}{L_y}\right) \sin\left(\frac{n_z \pi z}{L_z}\right) \quad (22)$$

$$E_{n_x, n_y, n_z} = \frac{\hbar^2 \pi^2}{2m} \left[ \left(\frac{n_x}{L_x}\right)^2 + \left(\frac{n_y}{L_y}\right)^2 + \left(\frac{n_z}{L_z}\right)^2 \right] \quad (23)$$

# Wave Functions



$$i\hbar \frac{\partial}{\partial t} \Psi(\mathbf{r}, t) = \hat{H} \Psi = \left( -\frac{\hbar^2}{2m} \nabla^2 + V(\mathbf{r}) \right) \Psi(\mathbf{r}, t) = -\frac{\hbar^2}{2m} \nabla^2 \Psi(\mathbf{r}, t) + V(\mathbf{r}) \Psi(\mathbf{r}, t)$$

$$V(r) = -\frac{1}{4\pi\epsilon_0} \frac{Ze^2}{r}$$

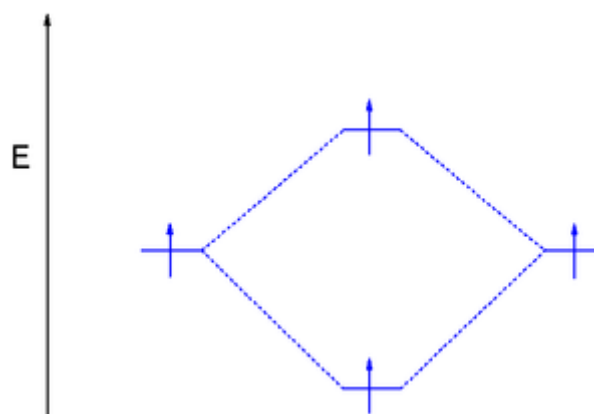
$$\psi_{n\ell m}(r, \vartheta, \varphi) = \sqrt{\left(\frac{2}{na_0}\right)^3 \frac{(n-\ell-1)!}{2n(n+\ell)!}} e^{-\rho/2} \rho^\ell L_{n-\ell-1}^{2\ell+1}(\rho) \cdot Y_\ell^m(\vartheta, \varphi)$$

# Linear combination of atomic orbitals molecular orbital method

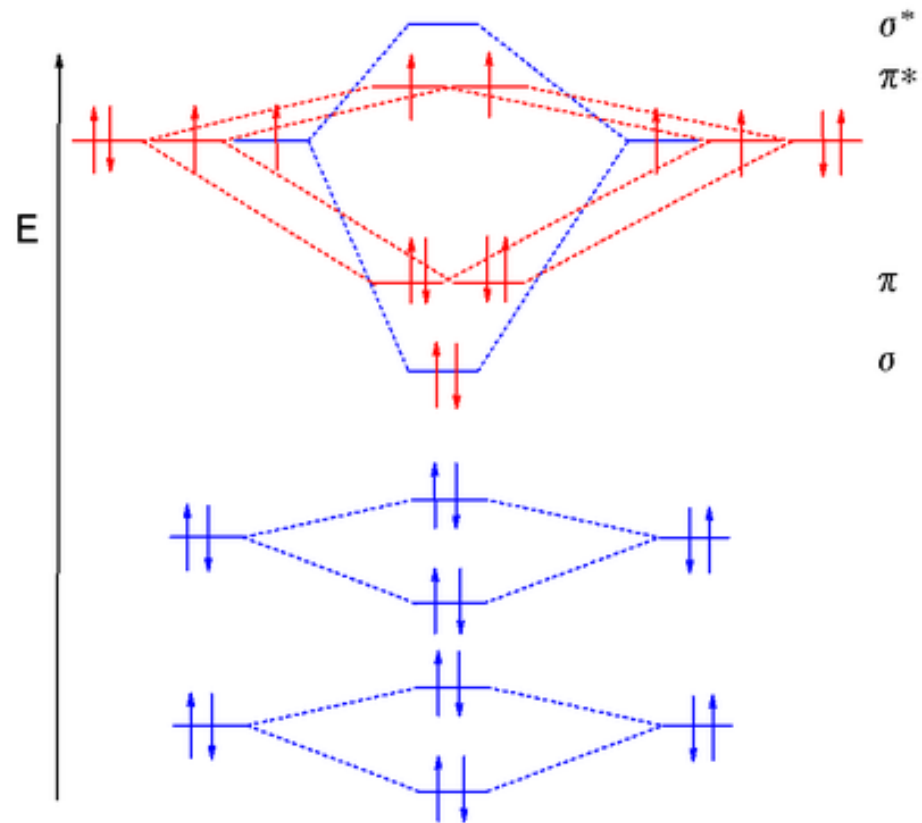
$$\phi_i = c_{1i}\chi_1 + c_{2i}\chi_2 + c_{3i}\chi_3 + \cdots + c_{ni}\chi_n$$

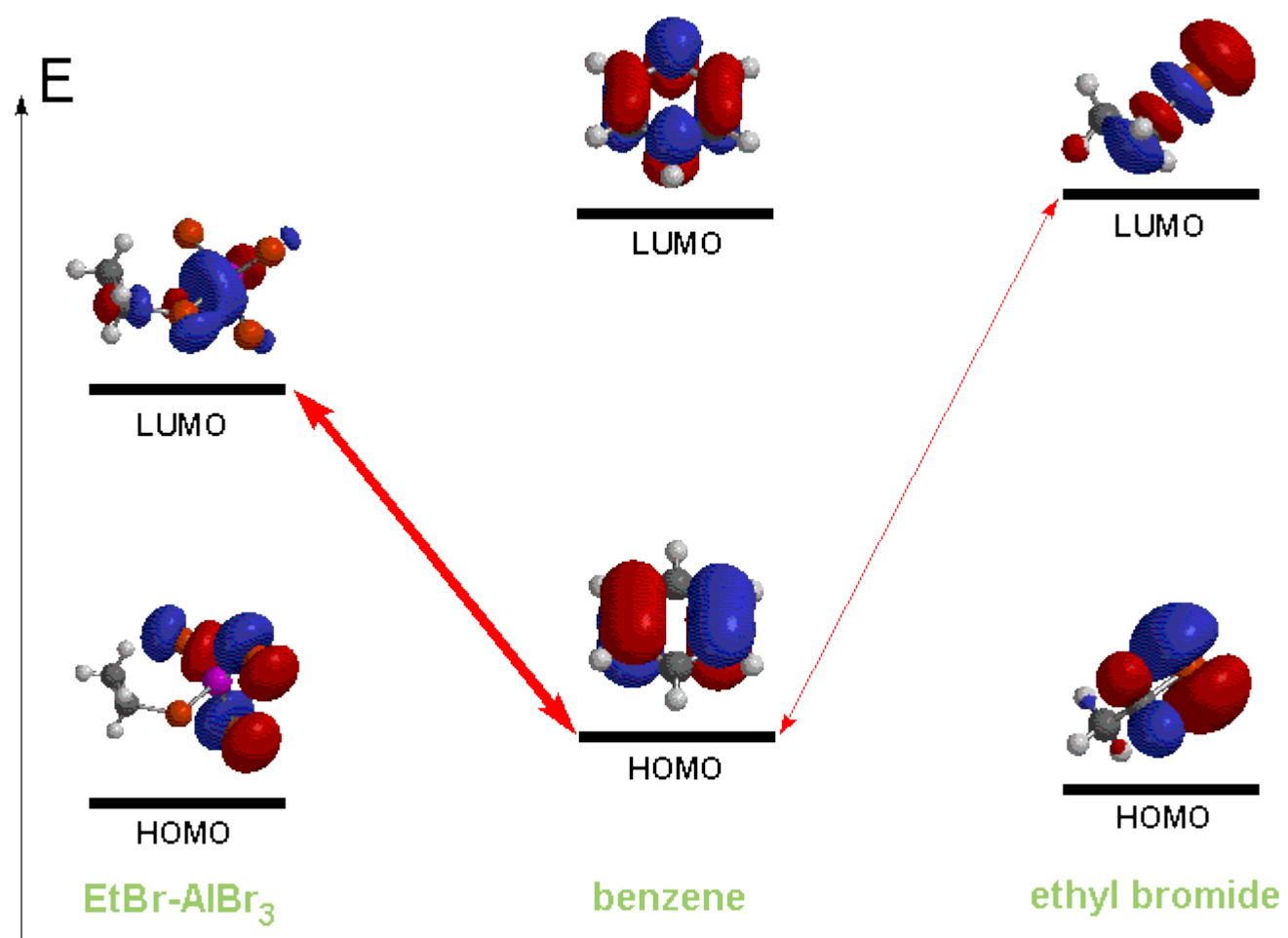
$$\psi_i = \sum_{\mu} c_{\mu i} \phi_{\mu}$$

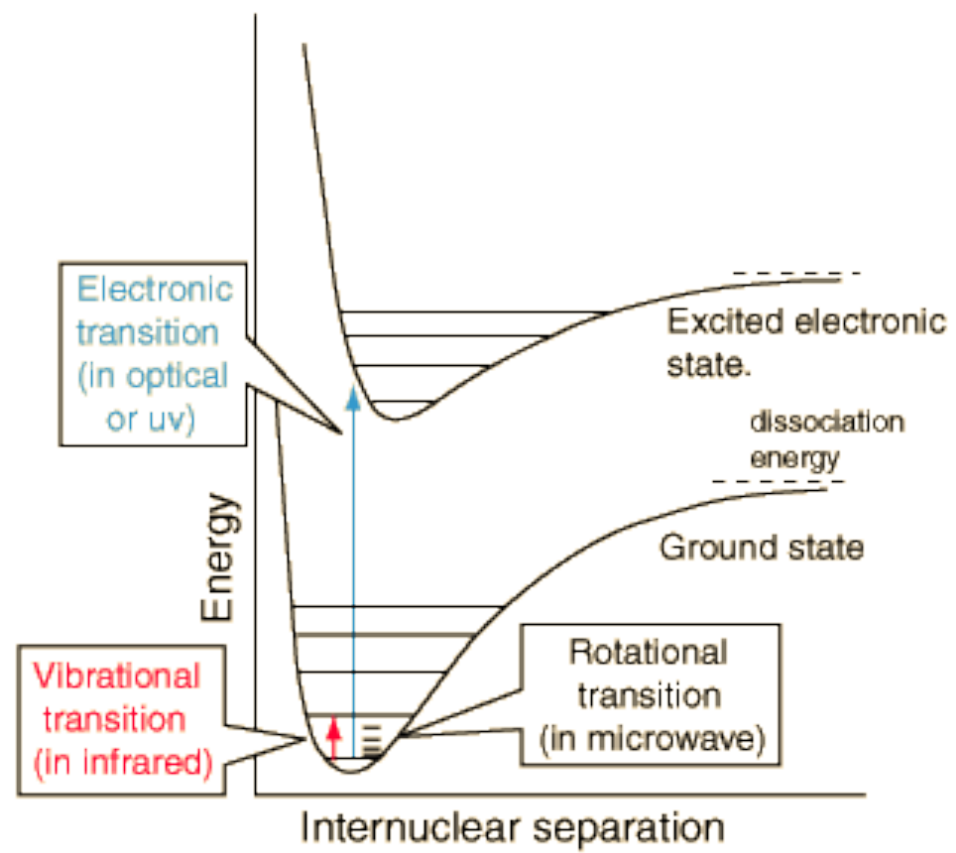
MO  $\nearrow$   $\nwarrow$  coefficient of  $\text{AO}_{\mu}$  in  $\text{MO}_i$   $\nearrow$  AO

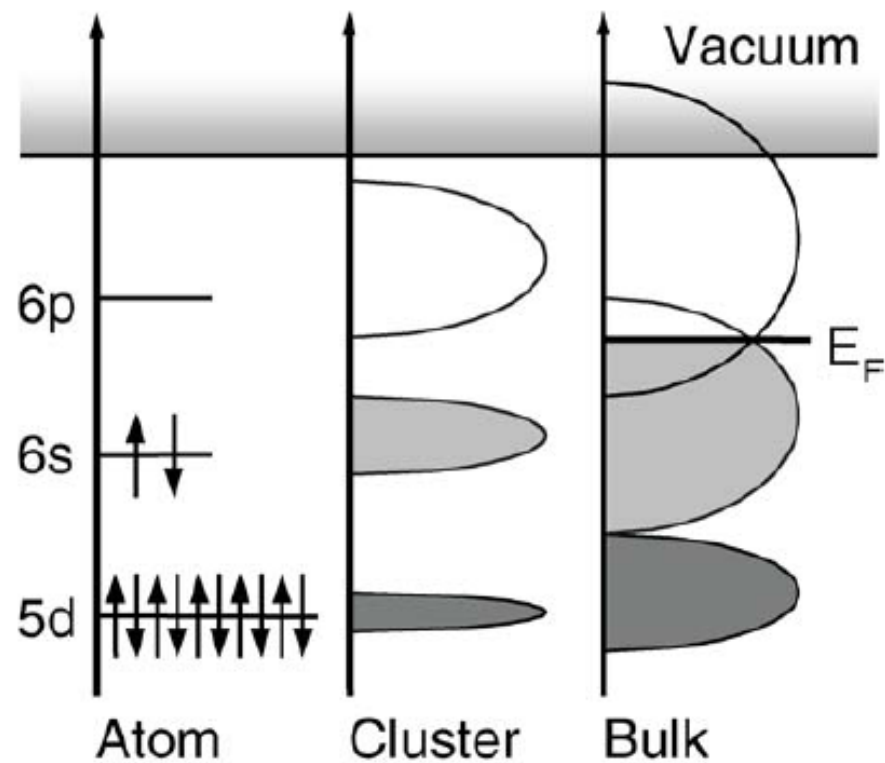


# Oxygen









**Figure 5** Energy diagram describing a generic Bloch-Wilson MIT in clusters (with specific reference to the energy levels of mercury). For sufficiently large clusters, the *s-p* band gap closes with increasing cluster size (shaded areas represent energy range with occupied electron levels). Overlap leads to a “continuous” DOS at  $E_F$  and to an Insulator to Metal transition.

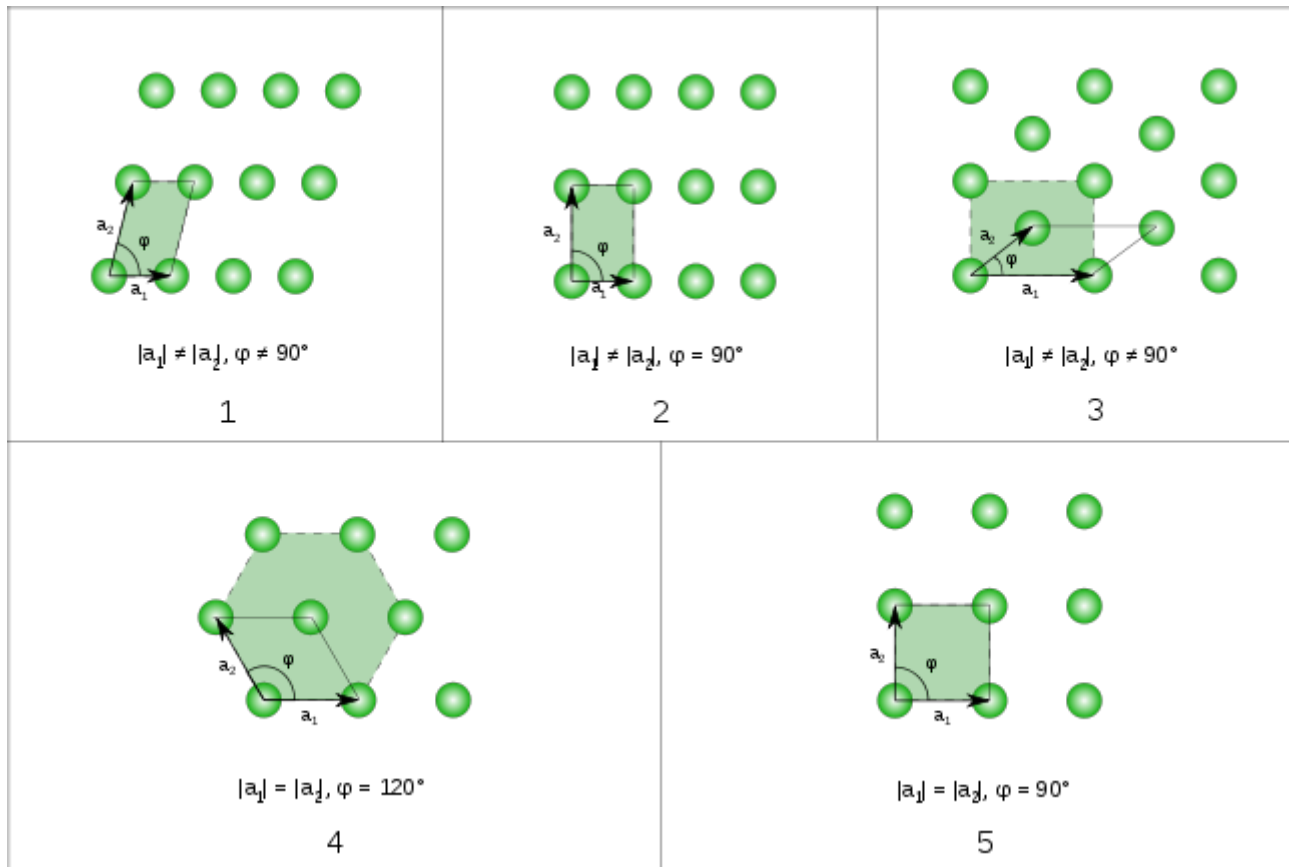
# Bloch wave

$$\psi_{n\mathbf{k}}(\mathbf{r}) = e^{i\mathbf{k}\cdot\mathbf{r}} u_{n\mathbf{k}}(\mathbf{r})$$

A **Bloch wave** or **Bloch state**, named after [Felix Bloch](#), is the [wavefunction](#) of a particle (usually, an [electron](#)) placed in a [periodic potential](#).

$$\epsilon_n(\mathbf{k}) = \epsilon_n(\mathbf{k} + \mathbf{K}),$$

# The five fundamental two-dimensional Bravais lattices



# Unit Cell

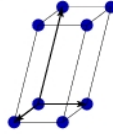
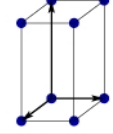
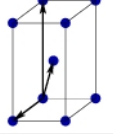
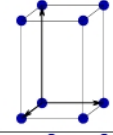
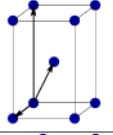
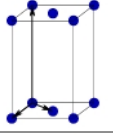
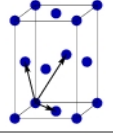
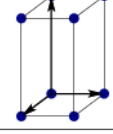
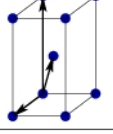
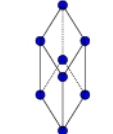
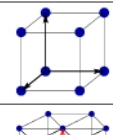
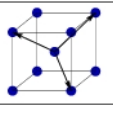
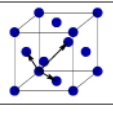
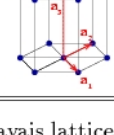
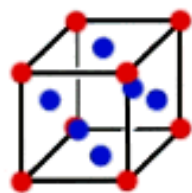
Bravais lattice	Parameters	Simple (P)	Volume centered (I)	Base centered (C)	Face centered (F)
Triclinic	$a_1 \neq a_2 \neq a_3$ $\alpha_{12} \neq \alpha_{23} \neq \alpha_{31}$				
Monoclinic	$a_1 \neq a_2 \neq a_3$ $\alpha_{23} = \alpha_{31} = 90^\circ$ $\alpha_{12} \neq 90^\circ$				
Orthorhombic	$a_1 \neq a_2 \neq a_3$ $\alpha_{12} = \alpha_{23} = \alpha_{31} = 90^\circ$				
Tetragonal	$a_1 = a_2 \neq a_3$ $\alpha_{12} = \alpha_{23} = \alpha_{31} = 90^\circ$				
Trigonal	$a_1 = a_2 = a_3$ $\alpha_{12} = \alpha_{23} = \alpha_{31} < 120^\circ$				
Cubic	$a_1 = a_2 = a_3$ $\alpha_{12} = \alpha_{23} = \alpha_{31} = 90^\circ$				
Hexagonal	$a_1 = a_2 \neq a_3$ $\alpha_{12} = 120^\circ$ $\alpha_{23} = \alpha_{31} = 90^\circ$				

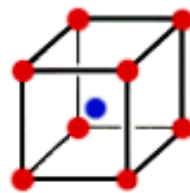
Table 1.1: Bravais lattices in three-dimensions.



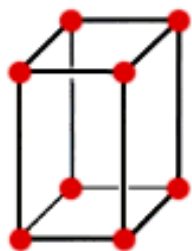
**Simple  
cubic**



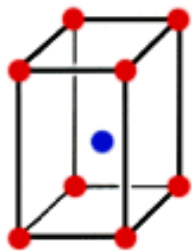
**Face-centered  
cubic**



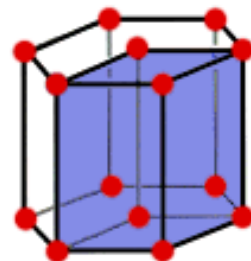
**Body-centered  
cubic**



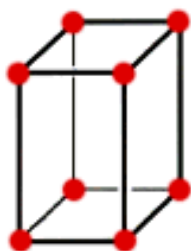
**Simple  
tetragonal**



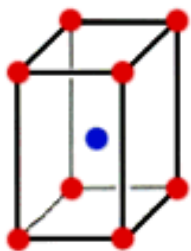
**Body-centered  
tetragonal**



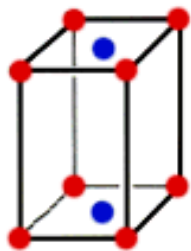
**Hexagonal**



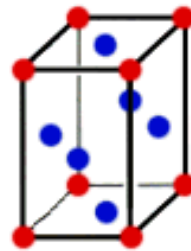
**Simple  
orthorhombic**



**Body-centered  
orthorhombic**



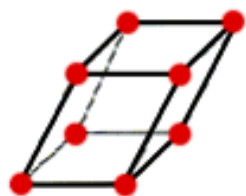
**Base-centered  
orthorhombic**



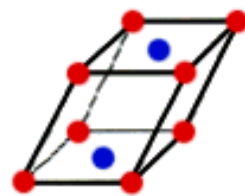
**Face-centered  
orthorhombic**



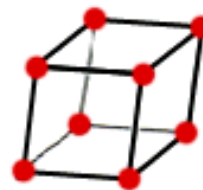
**Rhombohedral**



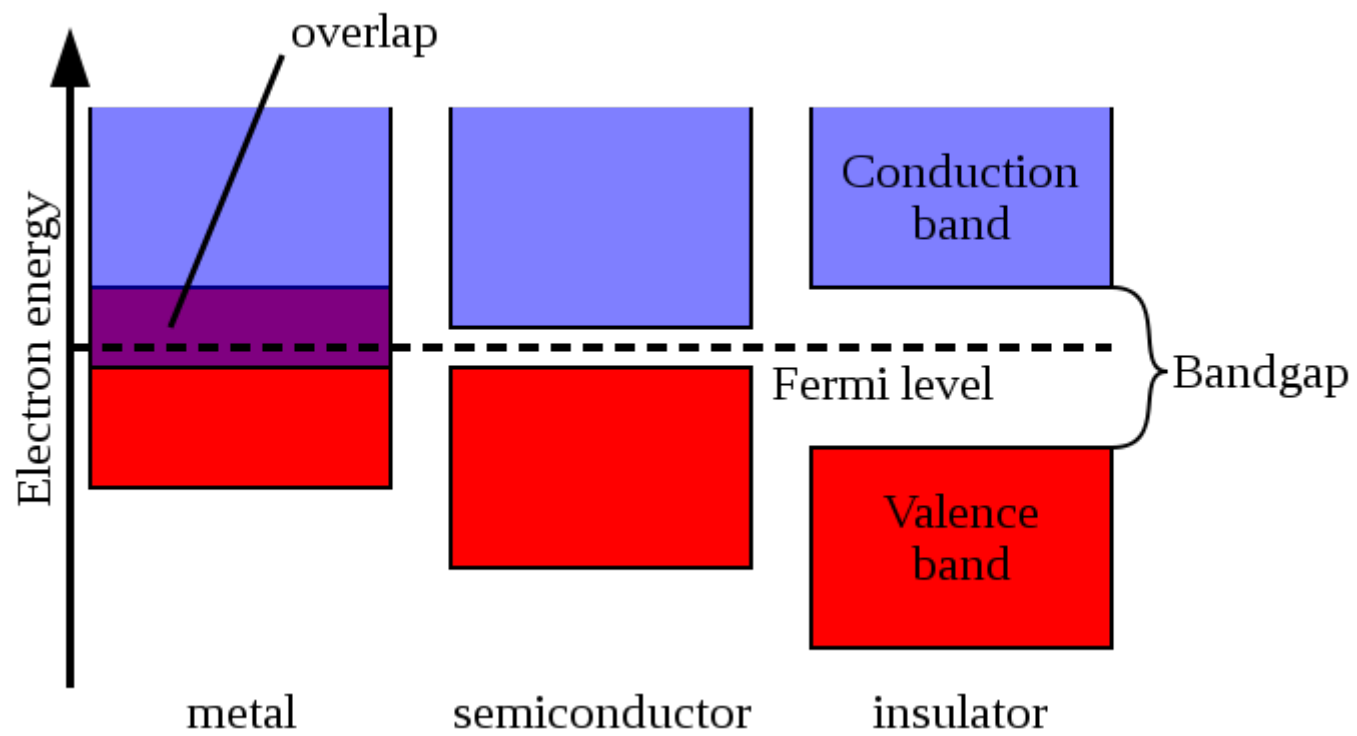
**Simple  
Monoclinic**



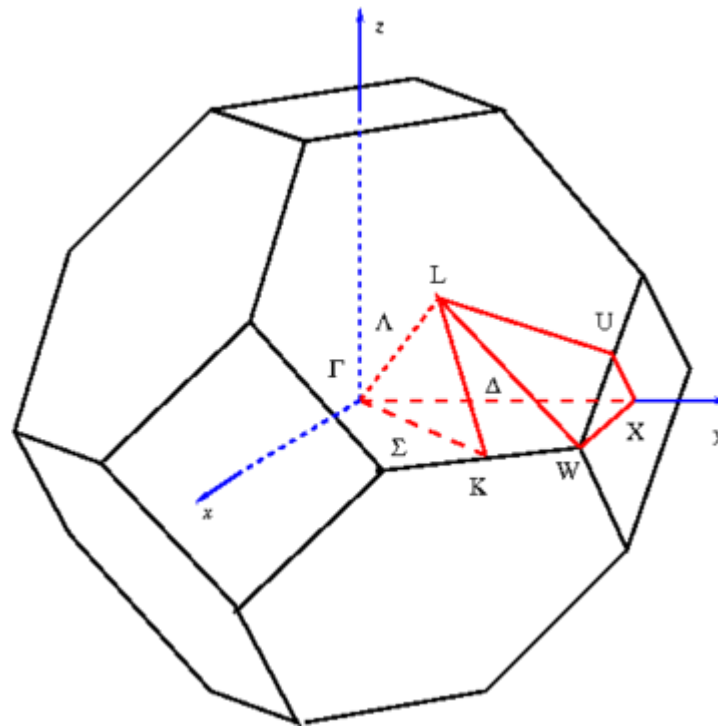
**Base-centered  
monoclinic**



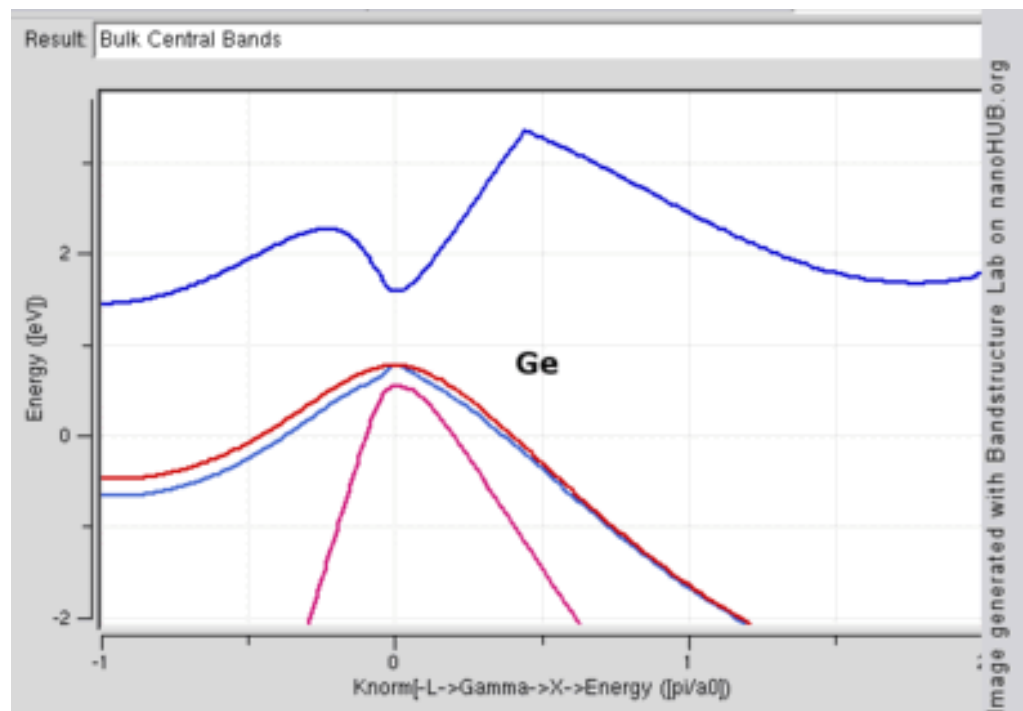
**Triclinic**



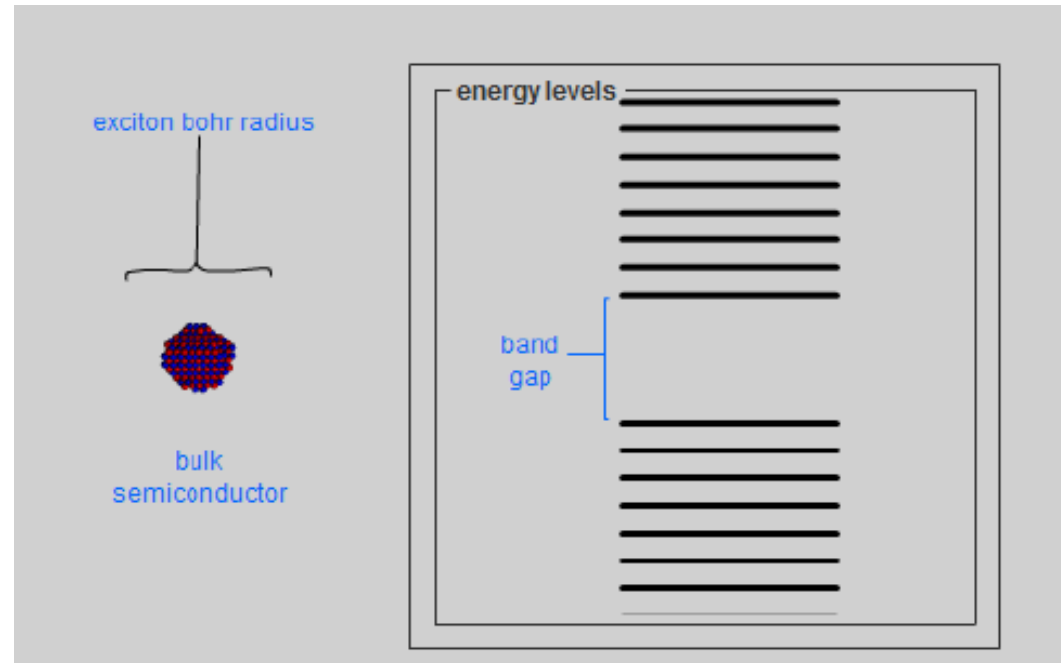
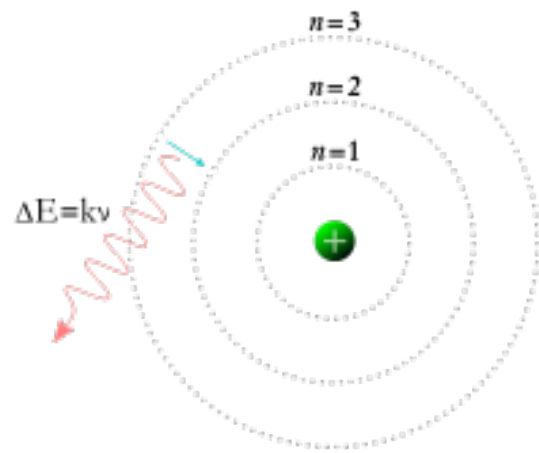
# First Brillouin zone of FCC lattice showing symmetry labels



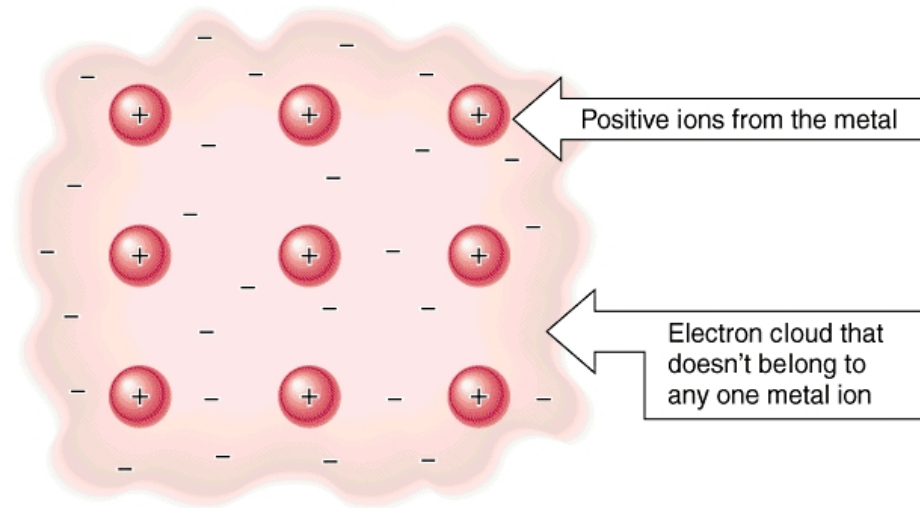
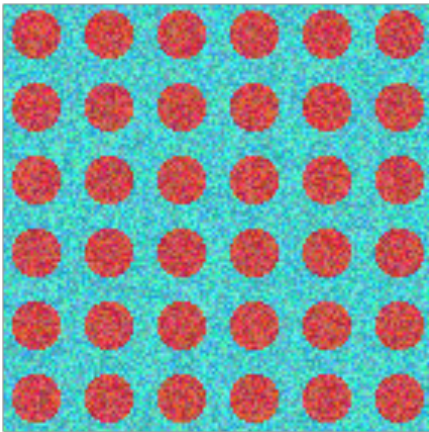
# Band Structures



# Bohr Exciton Radius



# Electron Sea

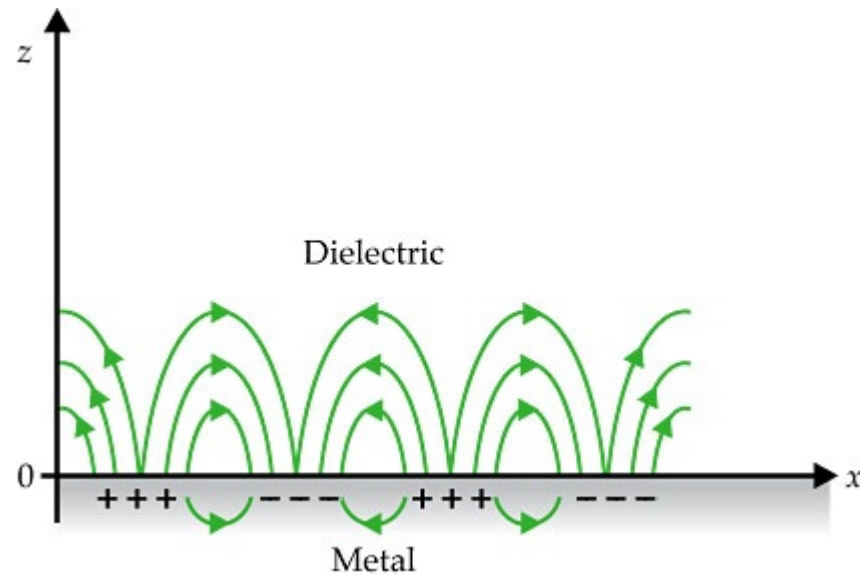


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$$m \frac{d^2 \delta x}{dt^2} = e E_x = -m \omega_p^2 \delta x,$$

$$\omega_p^2 = \frac{n e^2}{\epsilon_0 m},$$

# Surface Plasmon



$$\epsilon_m = 1 - \frac{\omega_p^2}{\omega^2}$$

# TiO<sub>2</sub>

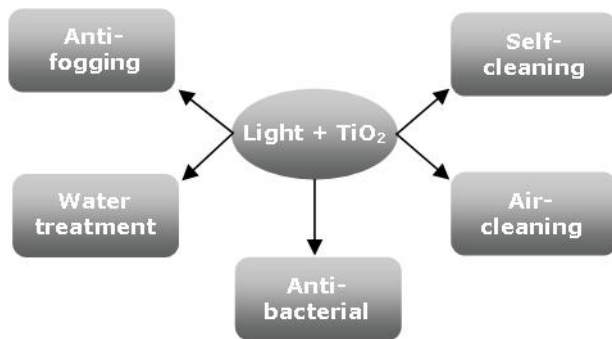
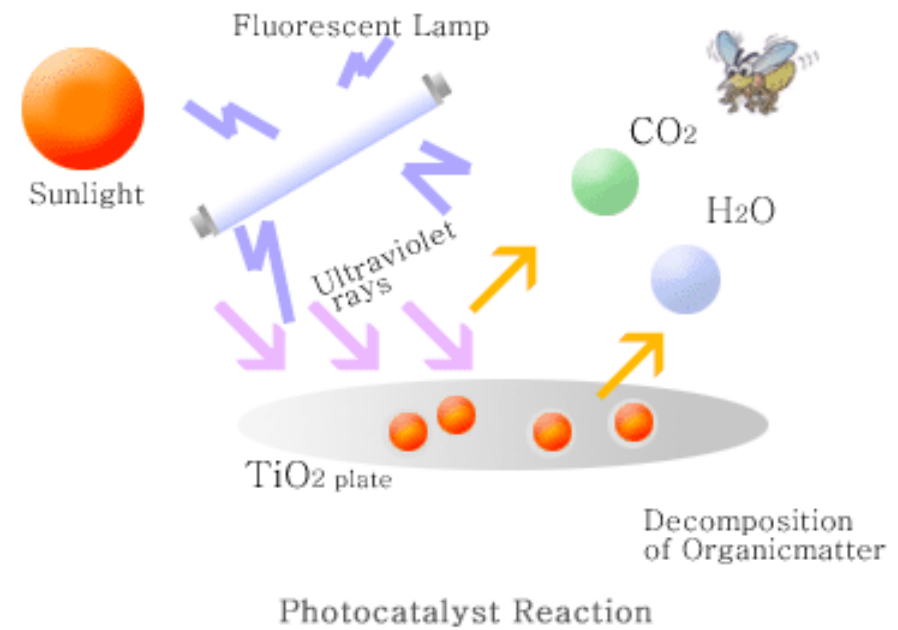
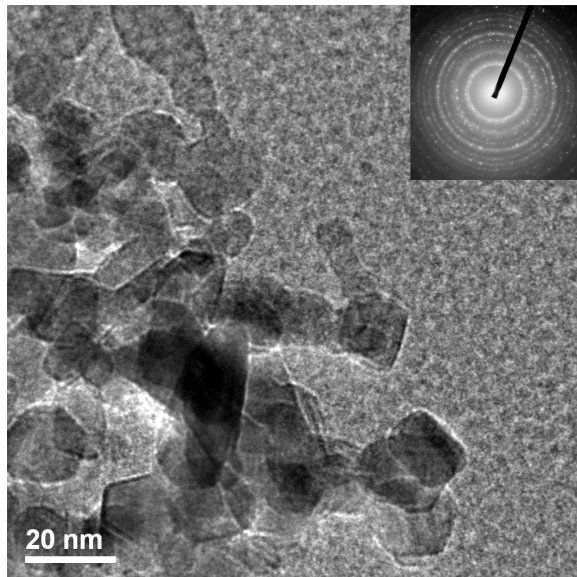
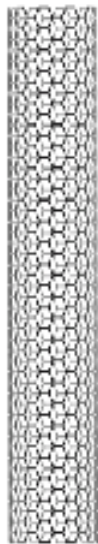


Figure 1. Major areas of activity in titanium dioxide photocatalysis



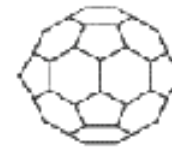
# Carbon



SWNT



Poly-C<sub>60</sub>

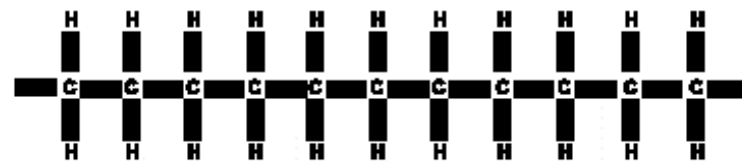
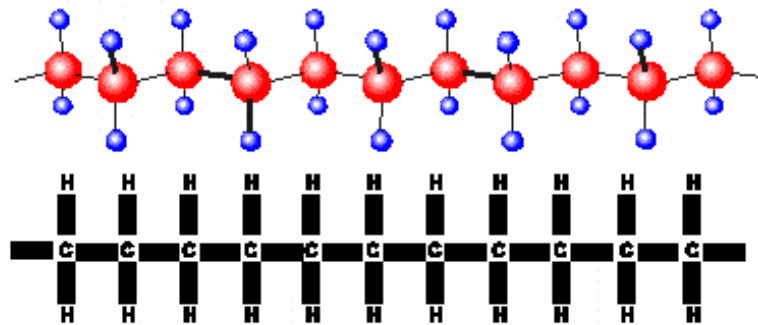
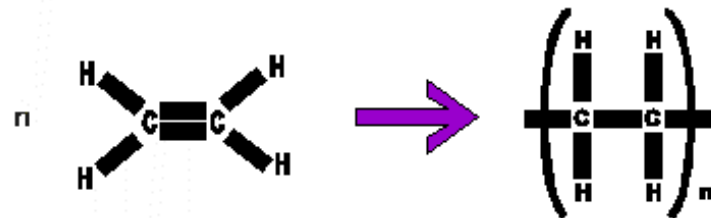
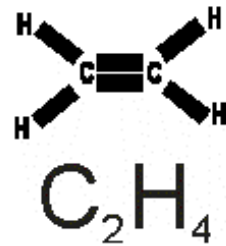
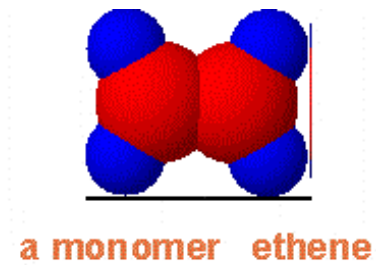


C<sub>60</sub>



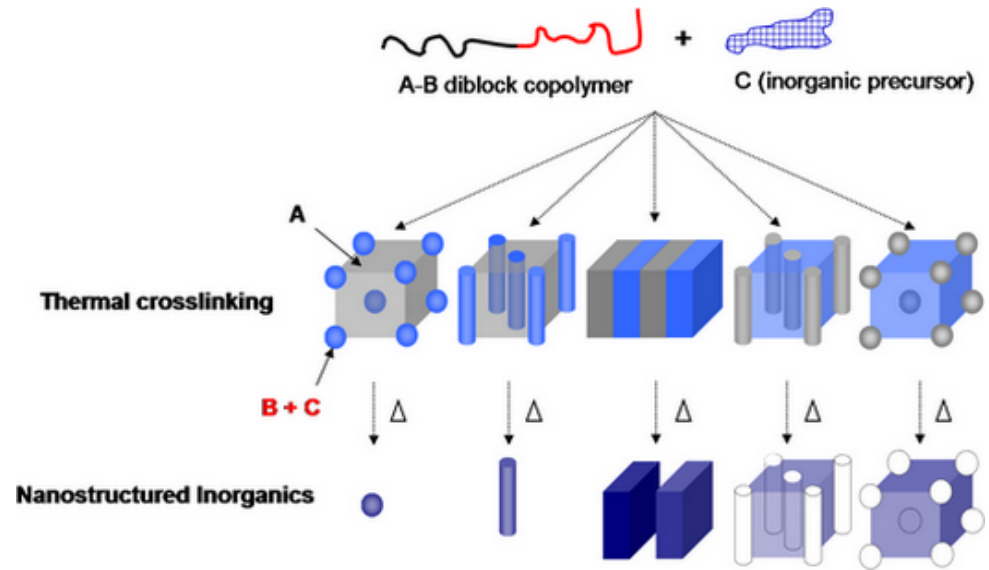
Nanodiamond  
~ 2-10 nm

# Polymer

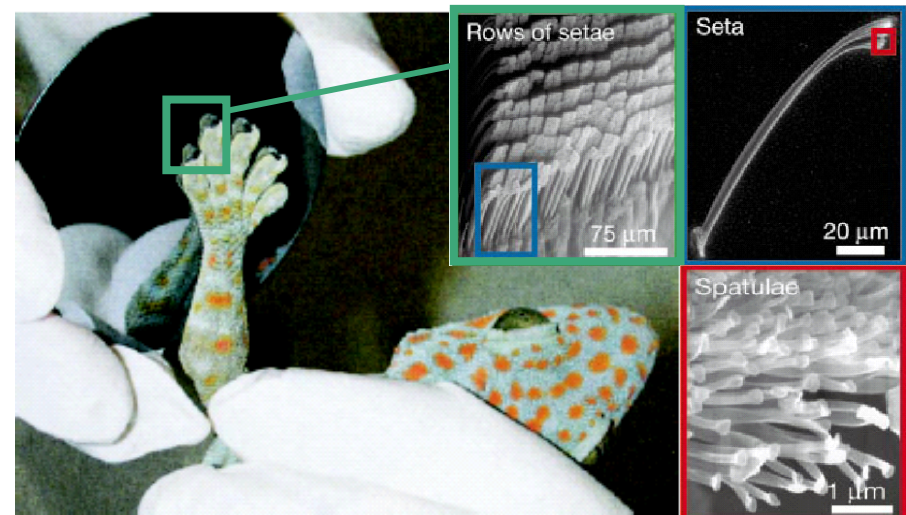
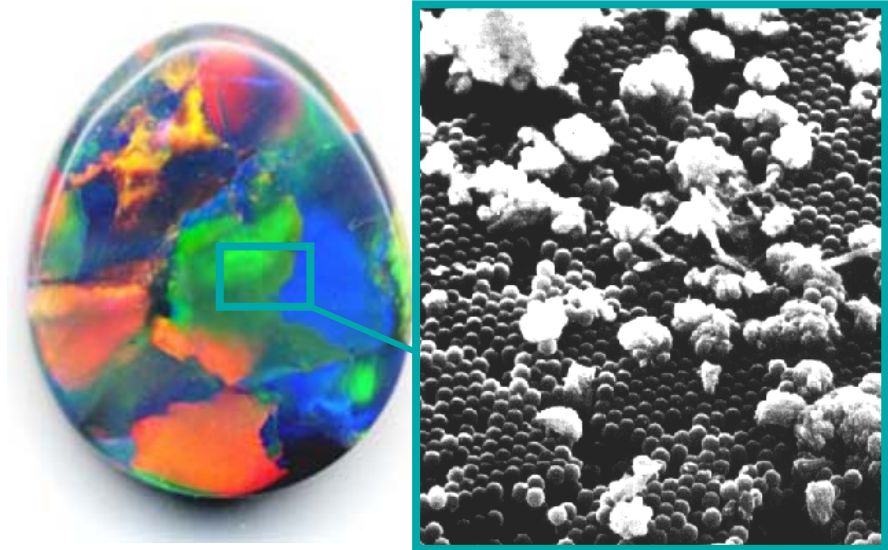
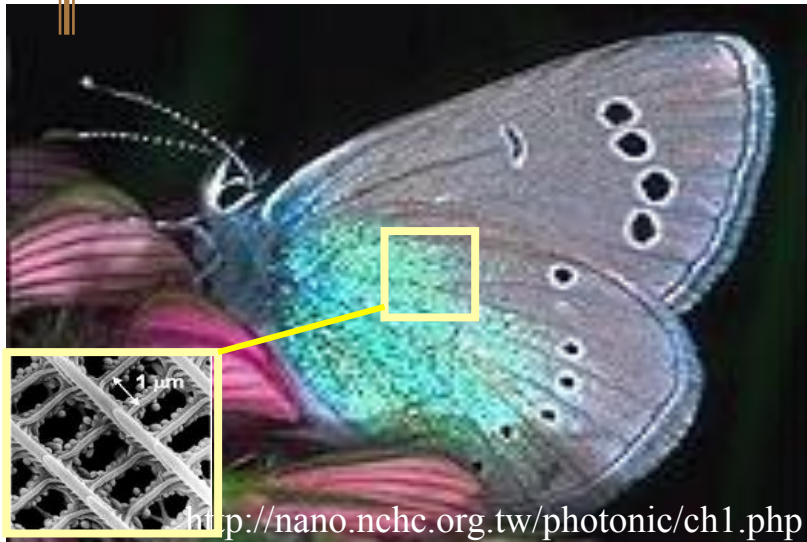


a polymer

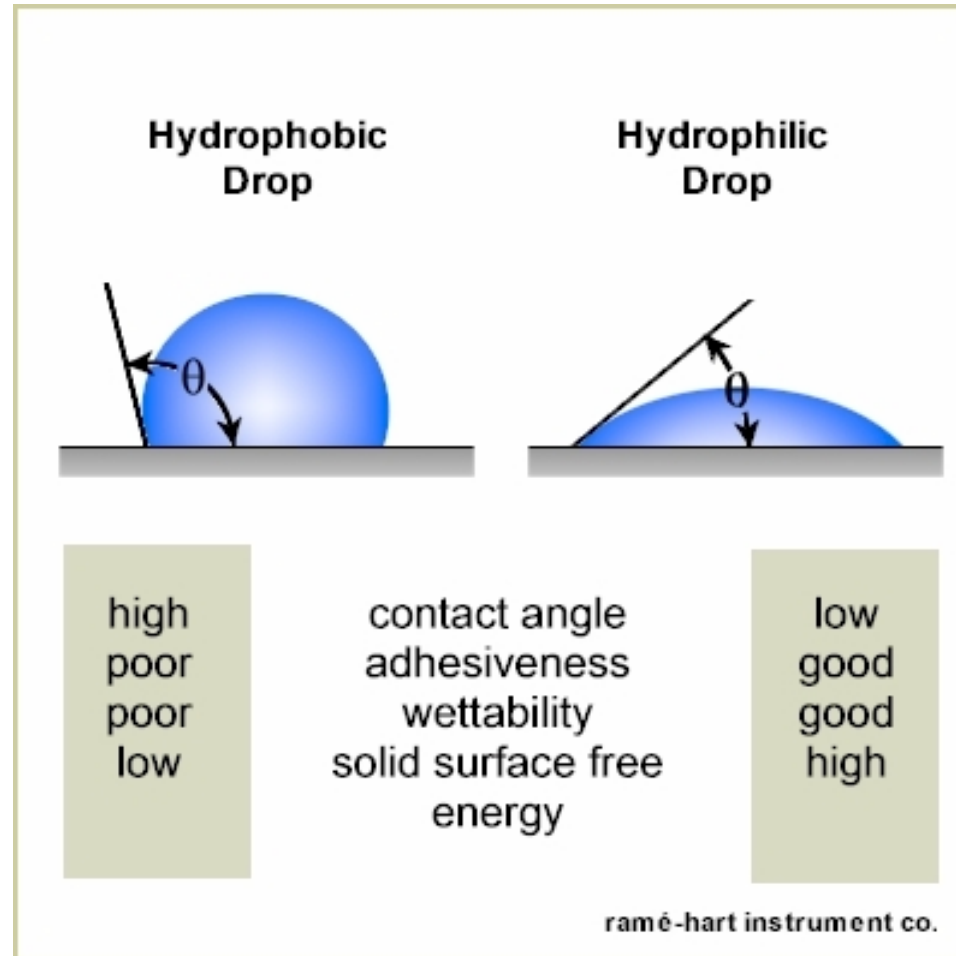
poly(ethene)



# Nature Materials

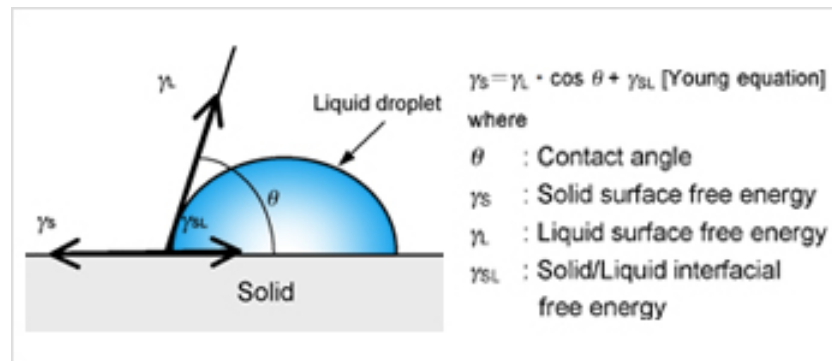


# Contact Angle



# Young's Equation

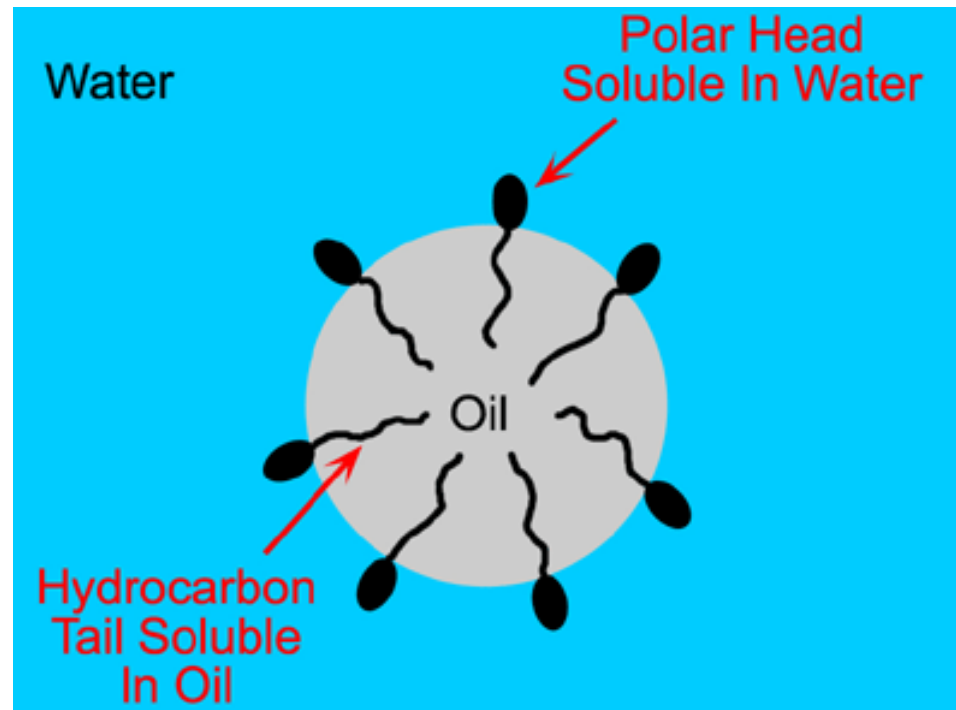
$$\gamma_{SL} + \gamma_{LV} \cos \theta_c = \gamma_{SV}$$



# Surface Energy Minimization

- Surfactants
- DLVO
- Polymeric
- Nucleation
- Ostwald Ripening
- Sintering
- Restructure

# Surfactant



# DLVO Theory

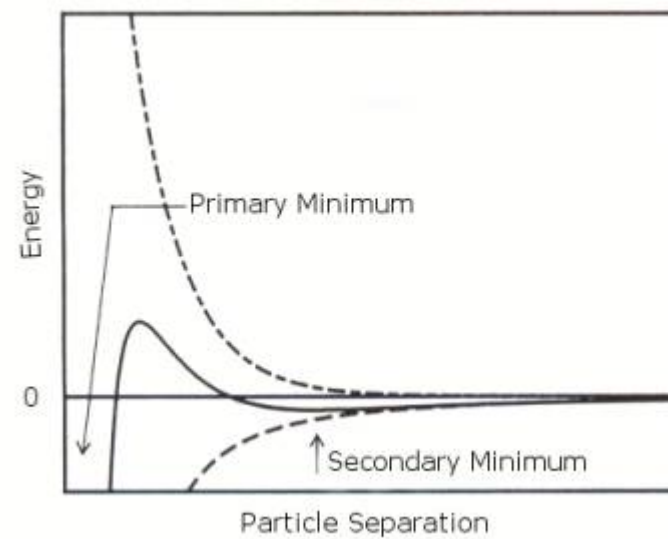
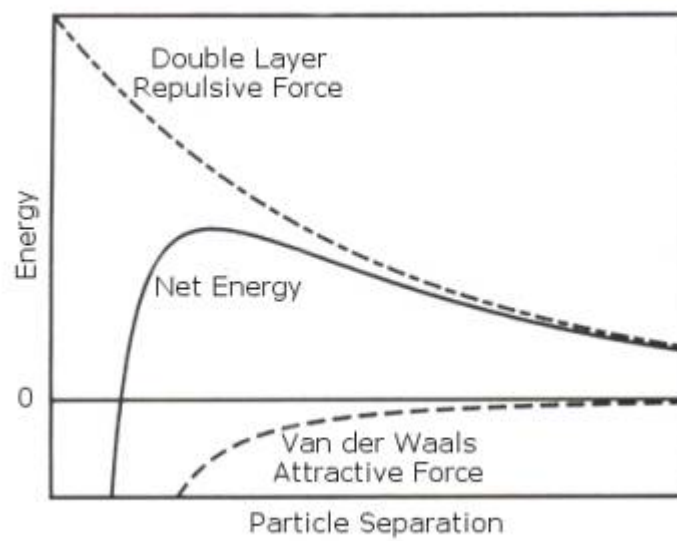
$$V_T = V_A + V_R + V_S$$

$$V_A = -A/(12 \pi D^2)$$

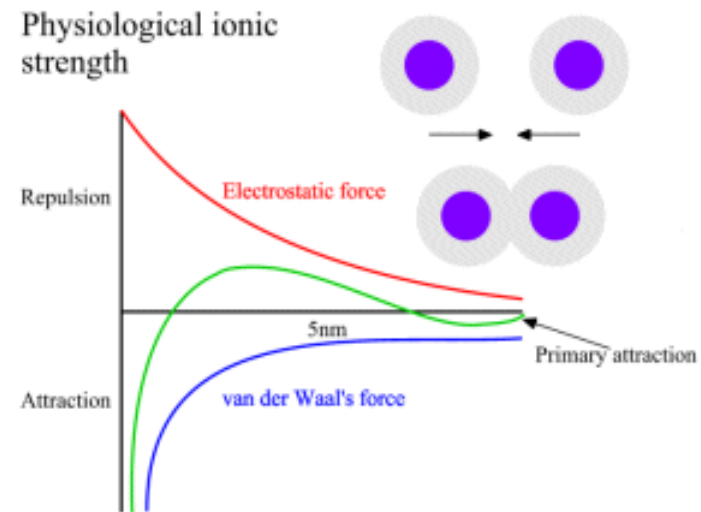
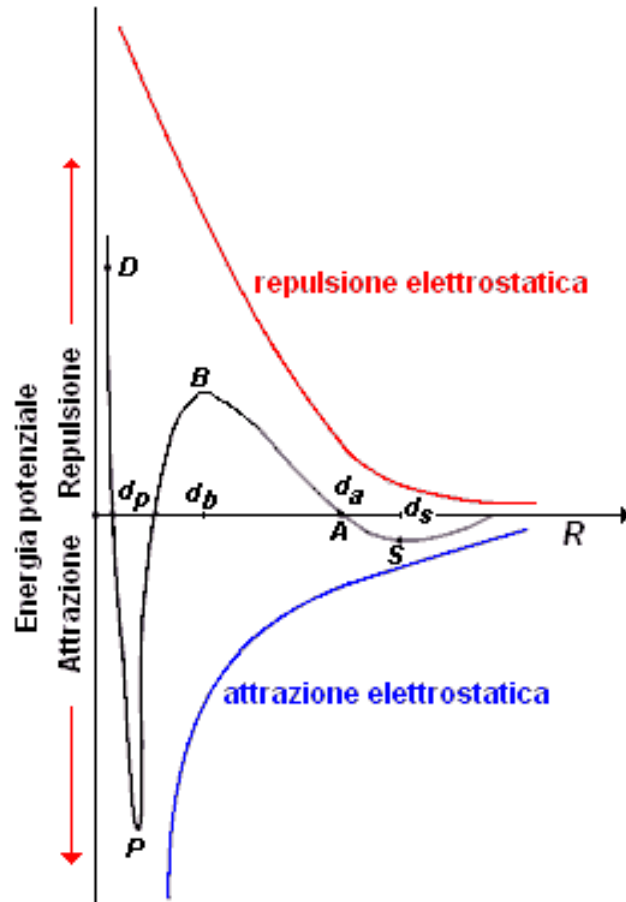
A is the Hamaker constant and D is the particle separation

$$V_R = 2 \pi \epsilon a \xi^2 \exp(-\kappa D)$$

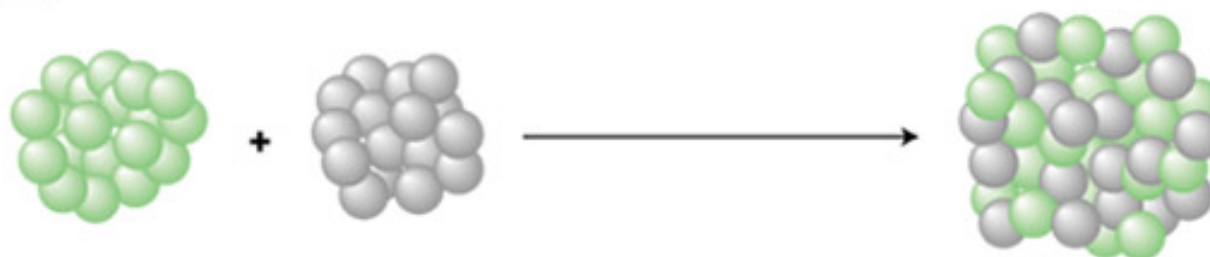
a is the particle radius,  $\pi$  is the solvent permeability,  
 $\kappa$  is a function of the ionic composition and  $\xi$  is the zeta potential



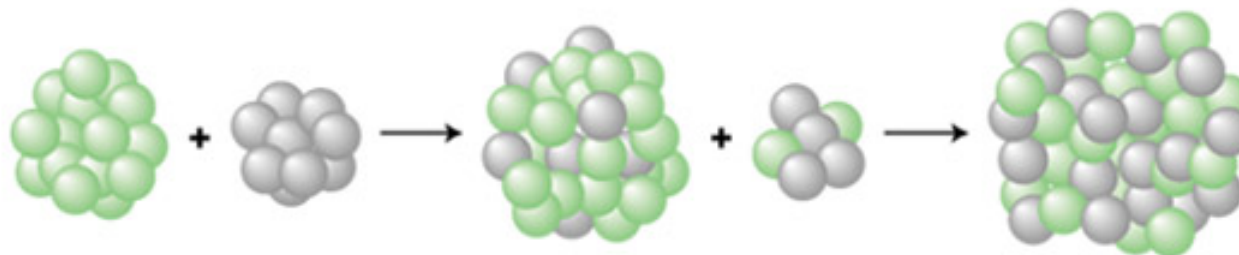
# DLVO Theory



**a** Coalescence



**b** Ostwald ripening



Two main mechanisms are shown here: **a**, coalescence sintering, and **b**, Ostwald ripening sintering. Coalescence sintering occurs when two clusters touch or collide and merge to form one bigger cluster. In contrast, Ostwald ripening sintering occurs by evaporation of atoms from one cluster, which then transfer to another. This is a dynamic process — both clusters exchange atoms, but the rate of loss from the smaller cluster is higher, because of the lower average coordination of atoms at the surface and their relative ease of removal. Thus big clusters get bigger at the expense of smaller clusters, which shrink and eventually disappear. The latter process is the usual form of sintering for metal clusters on a supported surface that are well spaced apart, although coalescence can occur for a high density of clusters. In general, the presence of the surface results in SMORS (surface-mediated Ostwald ripening sintering) in which material is transferred from one cluster to another by diffusion across the surface, and not through the gas phase.