

CMS II-6: Transition-metal oxides, strong correlations, and LDA+U

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Outline

- Insufficiencies of LDA
- How to improve LDA
- Self-interaction correction (SIC)
- LDA + Hubbard U (LDA+U) method
- Applications of LDA+U on various transition-metal oxides

Insufficiencies of LDA (LDA, LSDA, GGA)

- LDA is successful for lots of systems such as metallic or weakly correlated materials
- BUT LDA has some problems:
- Gaps too small or no gap,
PRB44, 943 (1991)
- Spin and orbital moment too small,
PRB44, 943 (1991)
- Especially for transition metal oxides or strongly correlated systems

TABLE II. Experimental (expt) and calculated (LDA + U) spin moments (m , in μ_B) and energy gaps (E , in eV) of the late-3d-transition-metal monoxides. For comparison, we also show these quantities as calculated from LSDA (Ref. 1).

	E_{LSD}	$E_{\text{LSD}+U}$	E_{expt}	m_{LSD}	$m_{\text{LSD}+U}$	m_{expt}
CaCuO ₂	0.0	2.1	1.5 ^a	0.0	0.66	0.51 ^b
CuO	0.0	1.9	1.4 ^c	0.0	0.74	0.65 ^d
NiO	0.2	3.1	4.3, ^e 4.0 ^f	1.0	1.59	1.77, ^g 1.64, ^h 1.90 ⁱ
CoO	0.0	3.2	2.4 ^{j,k}	2.3	2.63 (3.60)	3.35, ^l 3.8 ^m
FeO	0.0	3.2	2.4 ⁿ	3.4	3.62 (4.59)	3.32 ^m
MnO	0.8	3.5	3.6–3.8 ^o	4.4	4.61	4.79, ^g 4.58 ⁱ

^aY. Tokura, T. Arima, S. Koshihara, T. Ido, S. Ishibasi, H. Takagi, and S. Uchida, *Proceedings of the Second International Symposium on Superconductivity*, Tsukuba (Springer, New York, in press).

^bD. Vaknin, E. Cougnol, P. K. Davies, J. E. Fischer, D. C. Johnson, and D. P. Goshorn, *Phys. Rev. B* **39**, 9122 (1989).

^cF. P. Koffyberg and F. A. B

^dJ. B. Forsyth, P. J. Brown, a

^eG. A. Sawatzky and J. W. A

^fS. H fner, J. Osterwalder, T.

^gB. E. F. Fender, A. J. Jacob

^hH. A. Alperin, *J. Phys. Soc.*

ⁱA. K. Cheetham and D. A. C

^jJ. van Elp *et al.* (unpublished).

^kR. J. Powell and W. E. Spicer, *Phys. Rev. B* **2**, 2182 (1970).

^lD. C. Kahn and R. A. Ericson, *Phys. Rev. B* **1**, 2243 (1970).

^mW. L. Roth, *Phys. Rev.* **110**, 1333 (1958); D. Hermann-Ronzaud, P. Burlet, and J. Rossat Mignod, *J. Phys. C* **11**, 2123 (1978).

ⁿH. K. Bowen, D. Adler, and B. H. Auker, *J. Solid State Chem.* **12**, 355 (1975).

^oR. N. Iskenderov, I. A. Drabkin, L. T. Emel'yanova, and Ya. Ksendzov, *Fiz. Tverd. Tela* (Leningrad) **10**, 2573 (1968) [*Sov. Phys.—Solid State* **10**, 2031 (1969)]; L. Messick, W. C. Walker, and R. Glosser, *Phys. Rev. B* **6**, 3941 (1972).

Gaps too small or no gap
Spin and orbital moment too small
PRB 44 (1991) 943

Beyond LDA

- Self-interaction correction (SIC)
PRB23(1981)5048, PRL65(1990)1148
- Optimized effective potential method (OEP)
- Hartree-Fock (HF) method, PRB48(1993)5058
- GW approximation (GWA), PRB46(1992)13051,
PRL74(1995)3221
- Time-dependent density functional theory
(TDDFT)
- Dynamical mean field theory (DMFT)
- Quantum Monte-Carlo method (QMC)
- LDA+Hubbard U (LDA+U) method,
PRB44(1991)943, PRB48(1993)16929

Local density approximation (LDA)

Kohn-Sham scheme PR140(1965)A1133

$$E_G[n(\vec{r})] = T_S[n] + \int n(\vec{r}) V_{ext}(\vec{r}) d^3r \\ + \frac{1}{2} \int \frac{n(\vec{r}) n(\vec{r}')}{|\vec{r} - \vec{r}'|} d^3r d^3r' + E_{xc}[n]$$

$$V(\vec{r}) = V_{ext}(\vec{r}) + \int d^3r' \frac{n(\vec{r}')}{|\vec{r} - \vec{r}'|} + V_{xc}[n(\vec{r})]$$

Self-interaction correction (SIC)

Perdew and Zunger, PRB23(1981)5048

$$E_G^{SIC}[n] = E_G^{LDA}[n]$$

$$-\frac{1}{2} \sum_i \int d^3r d^3r' \frac{n_i(\vec{r}) n_i(\vec{r}')}{|\vec{r} - \vec{r}'|} - \sum_i E_{xc}[n_i]$$

$$V_i^{SIC}(\vec{r}) = V^{LDA}(\vec{r}) - \int d^3r' \frac{n_i(\vec{r}')}{|\vec{r} - \vec{r}'|} - V_{xc}[n_i(\vec{r})]$$

Basic idea of LDA+U

PRB 44 (1991) 943, PRB 48 (1993) 169

- Delocalized s and p electrons : LDA
- Localized (strongly correlated) d or f electrons : +U

using on-site d-d Coulomb interaction
(Hubbard-like term)

$$U \sum_{i \neq j} n_i n_j$$

instead of averaged Coulomb energy

$$UN(N-1)/2$$

When to use LDA+U

- Systems that LDA gives bad results
- Narrow band materials : $U \geq W$
- Localized electron systems
- Transition-metal oxides
- Strongly correlated materials
- Insulators (for semiconductors: use GWA)
-

Where to find U and J

- PRB 44 (1991) 943 : 3d atoms
- PRB 50 (1994) 16861 : 3d, 4d, 5d atoms
- PRB 58 (1998) 1201 : 3d atoms
- PRB 44 (1991) 13319 : Fe(3d)
- PRB 54 (1996) 4387 : Fe(3d)
- PRL 80 (1998) 4305 : Cr(3d)
- PRB 58 (1998) 9752 : Yb(4f)

Notes on using LDA+U

- The magnitude of U is difficult to calculate and measure accurately, the deviation could be typically as large as $\pm 1\text{eV}$
- For the same element, U depends also on the ionicity in different compounds: the higher ionicity, the larger U
- One thus varies U in a reasonable range to obtain better results
- One might vary U in a much larger range to see the effect of U (qualitatively)
- $\rightarrow$ Self-consistent LDA+U (much more difficult)

Various LDA+U methods

- Hubbard model in mean field approx. (HMF)
LDA+U : PRB 44 (1991) 943 (**WIEN2K**, **LMTO**)
- Approximate self-interaction correction (SIC)
LDA+U : PRB 48 (1993) 16929 (**WIEN2K**)
- Around the mean field (AMF) LDA+U :
PRB 49 (1994) 14211 (**WIEN2K**)
- **Rotationally invariant LDA+U** :
PRB 52 (1995) R5468 (**VASP4.6**, **LMTO**)
- Simplified rotationally invariant LDA+U :
PRB 57 (1998) 1505 (**VASP4.6**, **LMTO**)

Rotationally invariant LDA+U: PRB52(1995)R5468

$$E = E^{LDA} - \frac{1}{2}Un(n-1) + \frac{1}{2}J[n^{\uparrow}(n^{\uparrow}-1) + n^{\downarrow}(n^{\downarrow}-1)]$$

$$+ \frac{1}{2} \sum_{\{m\}, \sigma} \{ \langle mm'| V_{ee} | m' m'' \rangle n_{mm'}^{\sigma} n_{m'' m'''}^{-\sigma}$$

$$+ (\langle mm'| V_{ee} | m' m'' \rangle - \langle mm'| V_{ee} | m'' m' \rangle) n_{mm'}^{\sigma} n_{m'' m'''}^{\sigma} \}$$

$$V_{mm'}^{\sigma} = -U(n - \frac{1}{2}) + J(n^{\sigma} - \frac{1}{2}) + \sum_{m'' m'''} \{ \langle mm'| V_{ee} | m' m'' \rangle n_{m'' m'''}^{-\sigma}$$

$$+ (\langle mm'| V_{ee} | m' m'' \rangle - \langle mm'| V_{ee} | m'' m' \rangle) n_{m'' m'''}^{\sigma} \}$$

V_{ee} : Slater integral

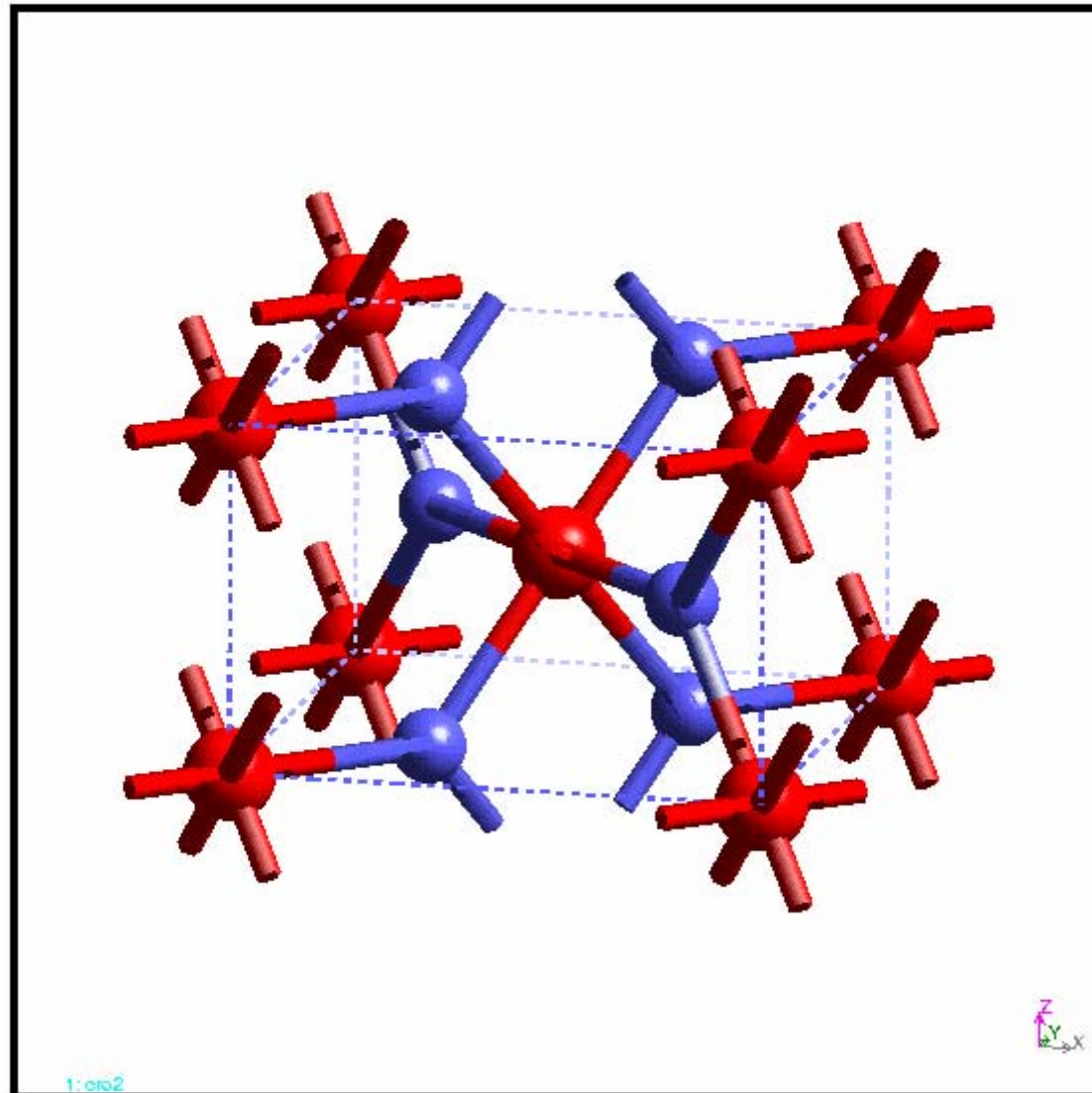
Applications of LDA+U on transition-metal oxides

- Rutile : CrO_2 (FP-LMTO)
- Pyrochlore : $\text{Cd}_2\text{Re}_2\text{O}_7$ (VASP)
- Double perovskite : $\text{Sr}_2\text{FeMoO}_6$, $\text{Sr}_2\text{FeReO}_6$, Sr_2CrWO_6 (FP-LMTO)
- Perovskite ruthenate SrRuO_3 (VASP)
- Cubic inverse spinel : high-temperature magnetite (Fe_3O_4) (FP-LMTO)
- Low-temperature charge-orbital ordering Fe_3O_4 (VASP)

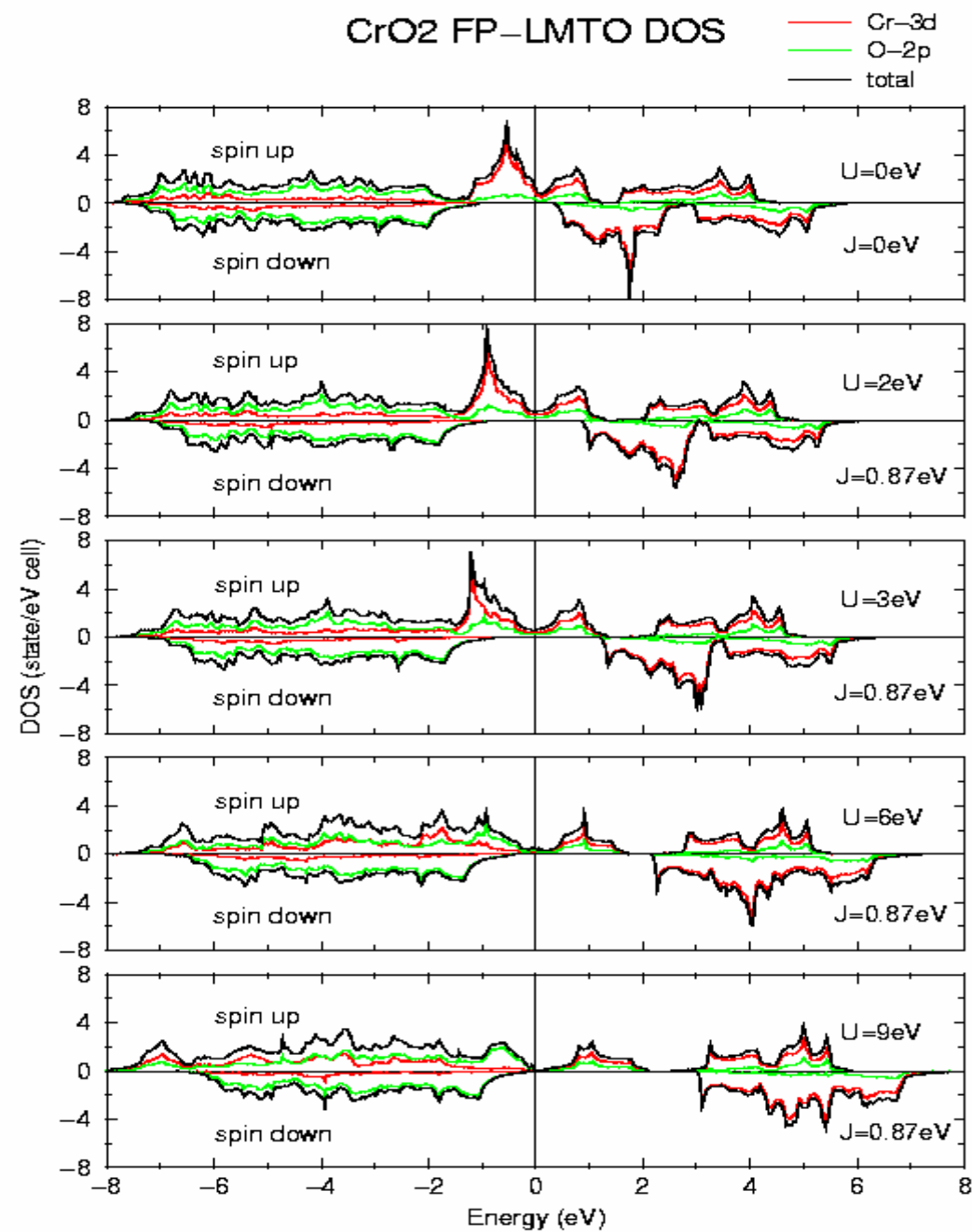
Rutile CrO₂

- Half-metal, moment = $2 \mu_B$
- Lattice type : bct
- 6 atoms in bct unit cell
- Space group : $P4_2/mnm$
- $a = 4.419 \text{ \AA}$, $c = 2.912 \text{ \AA}$, $u = 0.303$
- Ionic model : $\text{Cr}^{+4}(3d^2)$, $\text{O}^{-2}(2p^6)$
- $U = 3.0 \text{ eV}$, $J = 0.87 \text{ eV}$

Rutile structure



Jeng et al
JAP (2002)



PRB 56 (1997) 15509

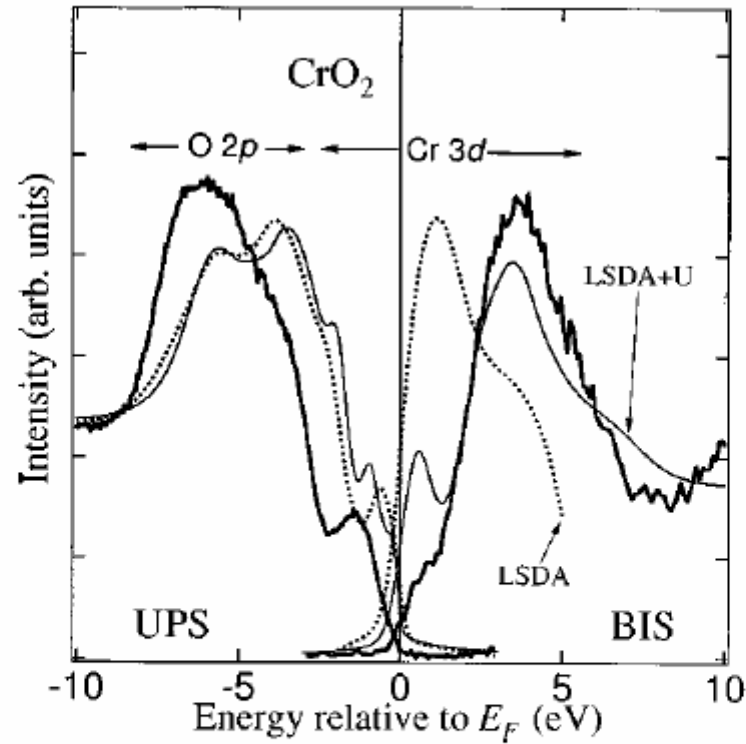


FIG. 4. UPS ($h\nu=40.8$ eV) and BIS ($h\nu=1486.6$ eV) spectra (solid curves) compared with theoretical spectra deduced from the LSDA (Ref. 3) and LSDA+ U (Ref. 8) calculations (dashed curves).

Spin and orbital magnetic moments of CrO₂

PRB66(2002)174440

(u _B)	Spin moment			Orbital moment	
	Cr	O	Total	Cr	O
LDA	1.89	-0.042	2.00	-0.037	-0.0011
LDA+U	1.99	-0.079	2.00	-0.051	-0.0025
Exp.			2	-0.05*	-0.003*

* D. J. Huang et al, SRRC

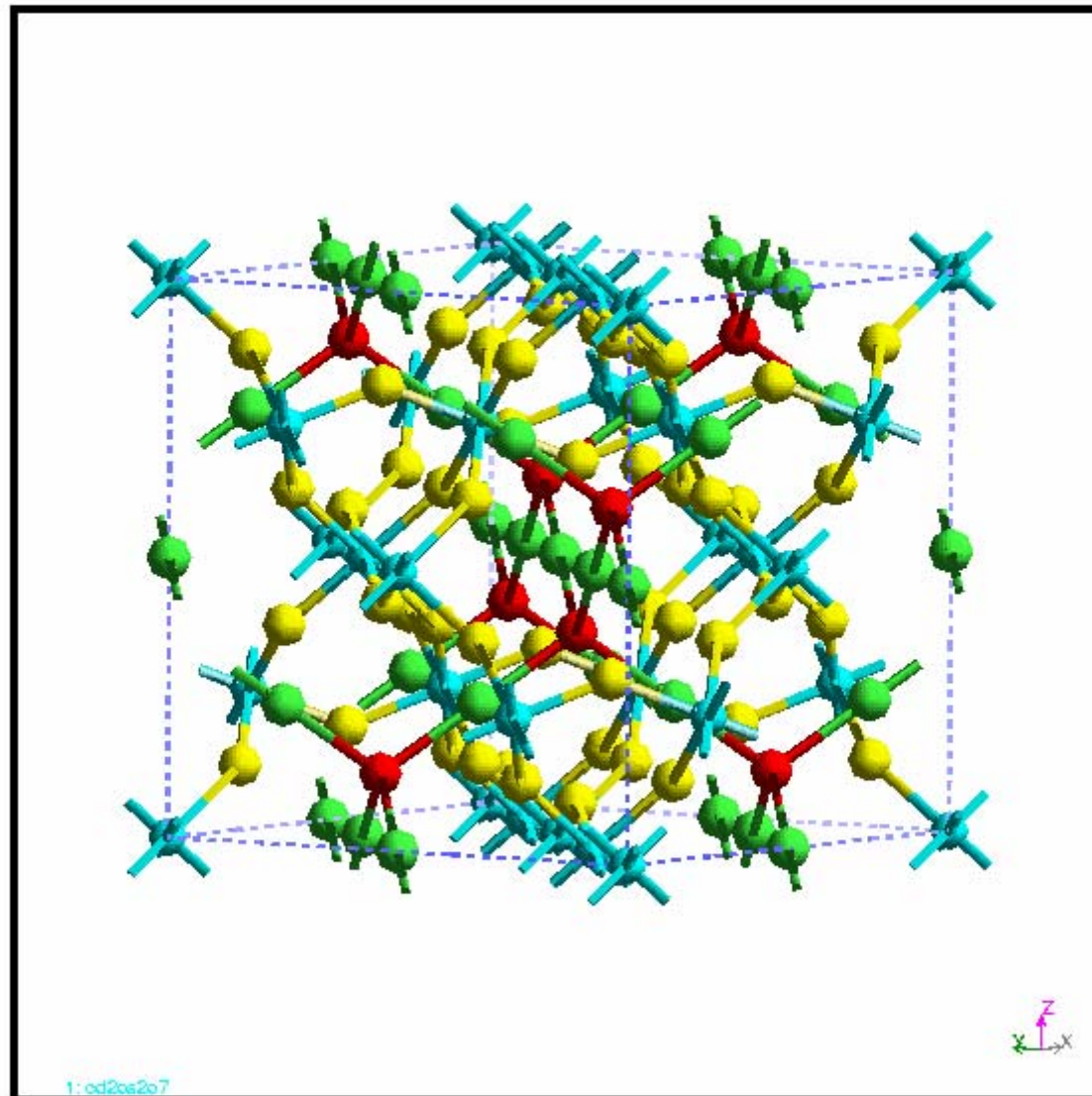
Rutile CrO₂

- LDA+U enhances the gap and the exchange splitting at the Fermi level
- LDA+U also gives larger spin and orbital magnetic moment
- $U \sim W$, orbital moment quenched,
→ stronger hybridization, stronger crystal field, → close to itinerant picture

Pyrochlore $\text{Cd}_2\text{Re}_2\text{O}_7$

- Lattice type : fcc
- 88 atoms in cubic unit cell
- Space group : $\text{Fd}\bar{3}\text{m}$
- $a = 10.219\text{\AA}$, $x=0.316$
- Ionic model: $\text{Cd}^{+2}(4\text{d}^{10})$, $\text{Re}^{+5}(5\text{d}^2)$, $\text{O}^{-2}(2\text{p}^6)$
- $U(\text{Cd}) = 5.5\text{ eV}$, $U(\text{Re}) = 3.0\text{ eV}$
- $J = 0\text{ eV}$ (no spin moment)

Pyrochlore structure



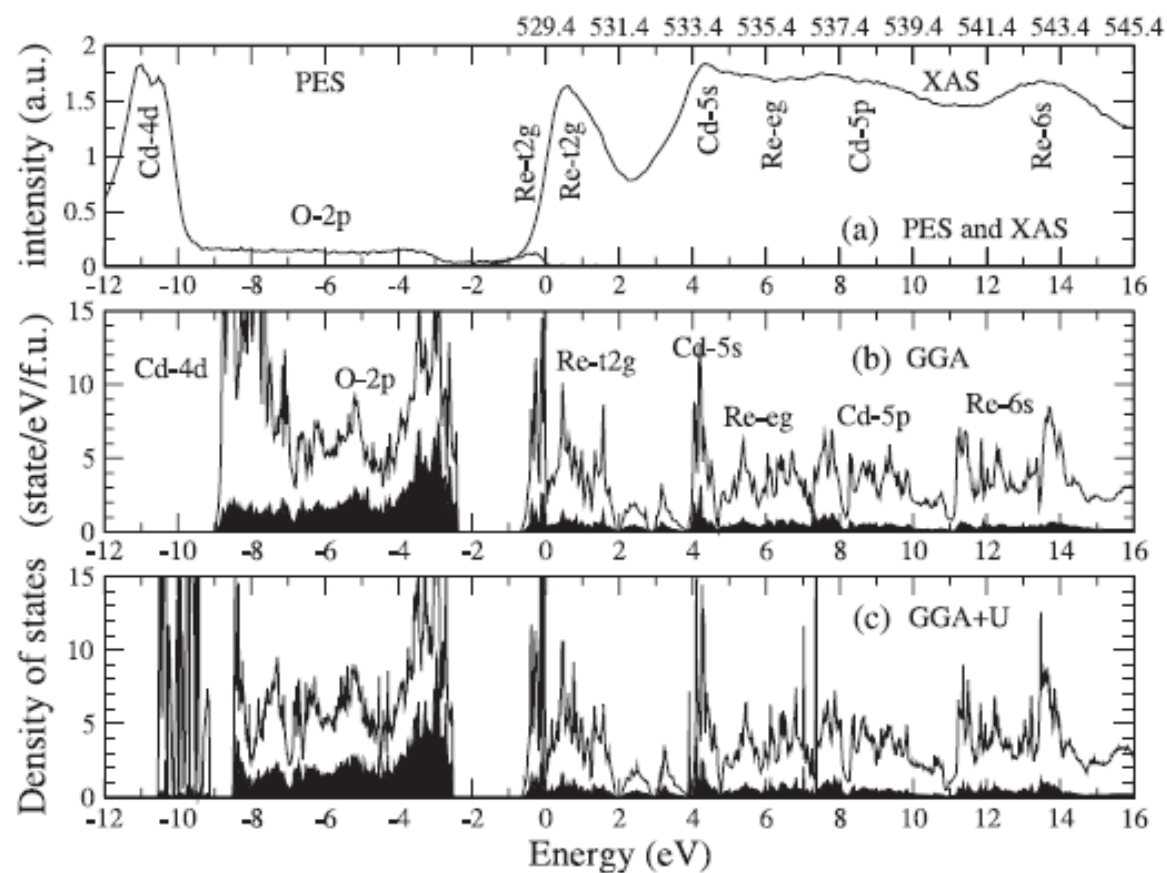


Figure 2. (a) PES [26] and O 1s XAS (this work) spectra of $\text{Cd}_2\text{Re}_2\text{O}_7$. The Fermi level of XAS is at 529.4 eV (reflection point method). (b) Total density of states from GGA (b) and GGA + U (c). The dark area in (b) and (c) are the contributions from O p states.

Pyrochlore $\text{Cd}_2\text{Re}_2\text{O}_7$

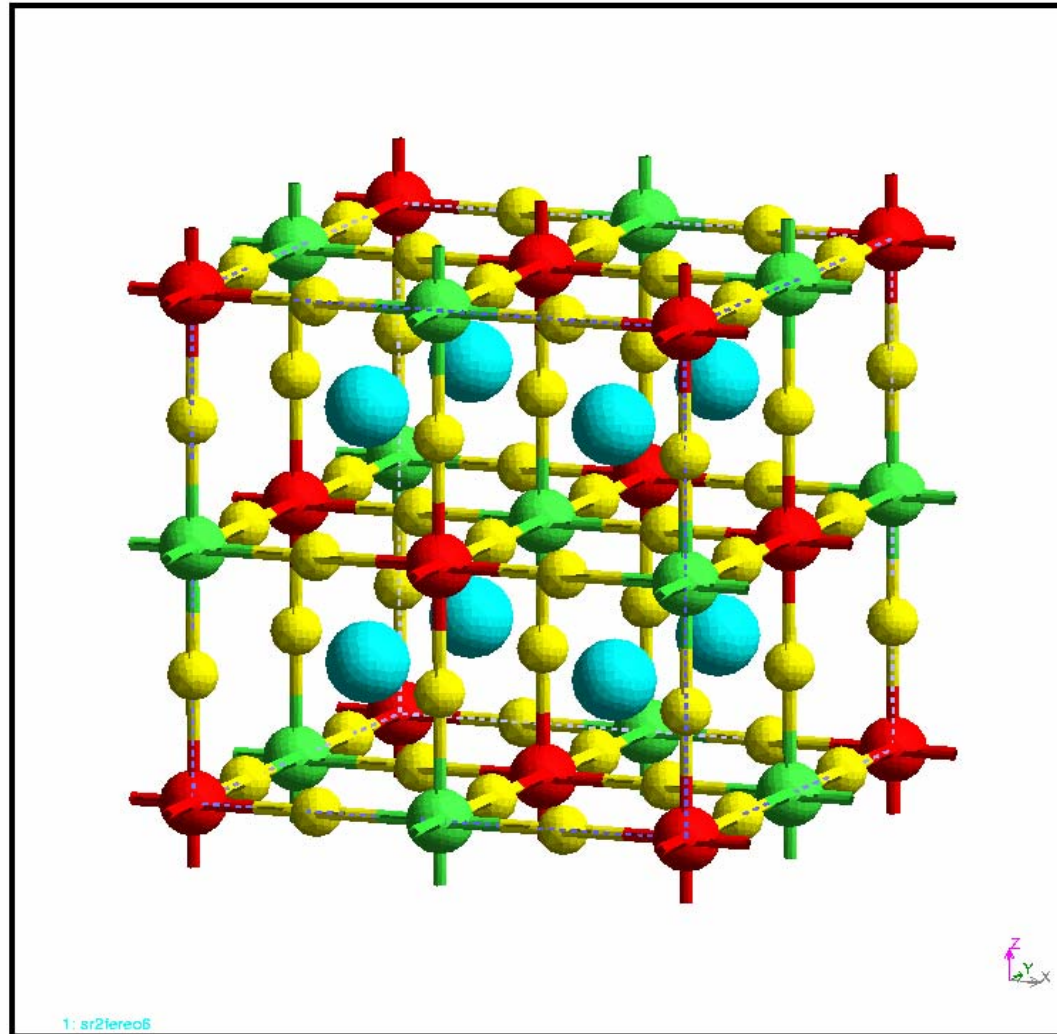
- LDA unoccupied DOS agree well with XAS data from K. D. Tsuei, SRRC
- Cd-4d band from LDA is ~ 3 eV higher than photo emission spectrum (PRB66(2002)1251)
- Cd-4d band from LDA+U agree well with photo emission spectrum (PRB66(2002)1251)
- Cd-4d orbital is close to localized electron picture, whereas the other orbitals are more or less itinerant

Double perovskites :

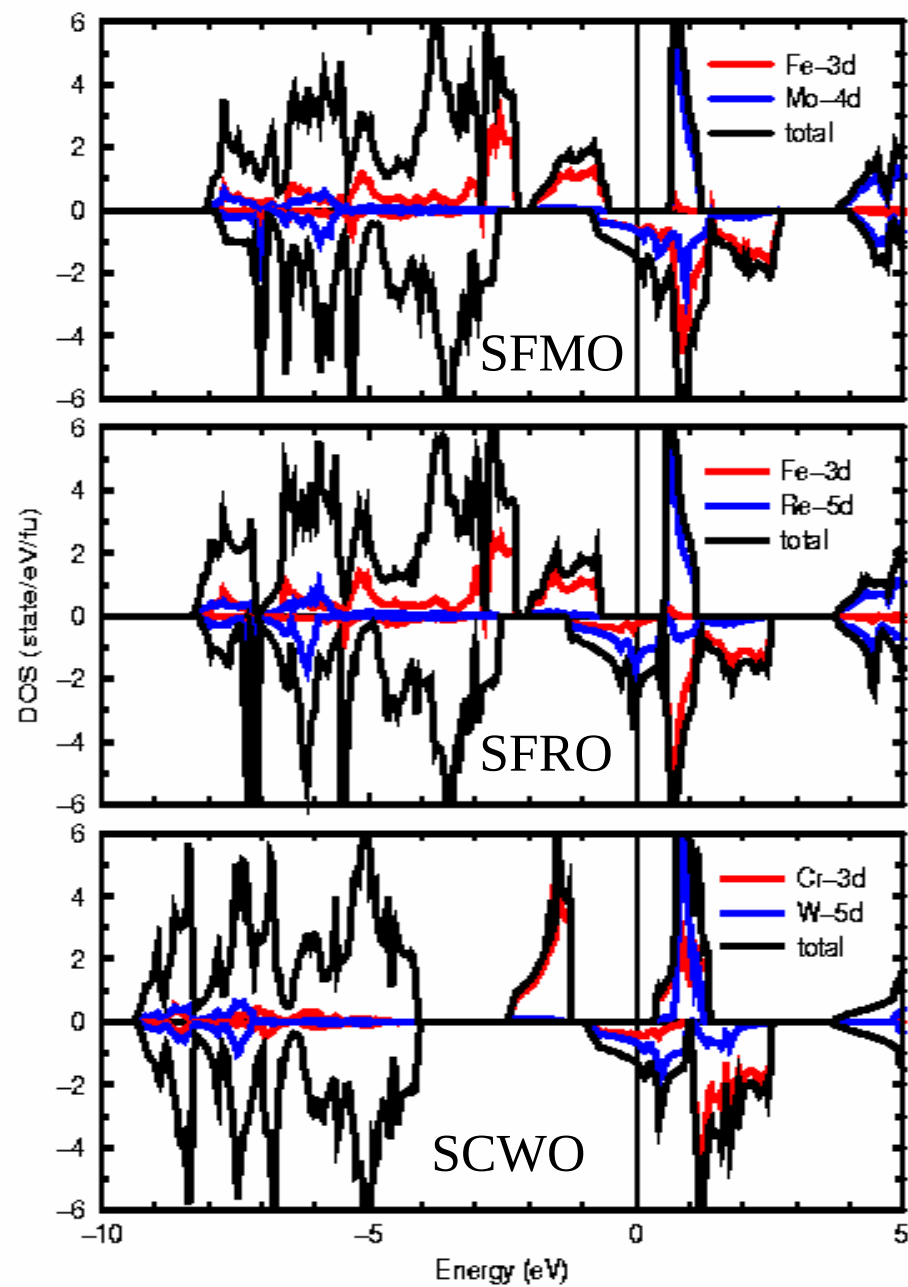
$\text{Sr}_2\text{FeMoO}_6$, $\text{Sr}_2\text{FeReO}_6$, Sr_2CrWO_6

- Half-metal, moment = 4, 3, 2 μ_B
- Lattice type : tet, fcc, fcc
- 40 atoms in tet, fcc, fcc unit cell
- Space group : $I4/mmm$, $Fm3m$, $Fm3m$
- $a = 7.89, 7.832, 7.878 \text{ \AA}$, $c/a = 1.001, 1, 1$
- Ionic model : $\text{Fe}^{+3}(3d^5)$, $\text{Cr}^{+3}(3d^3)$,
 $\text{Mo}^{+5}(4d^1)$, $\text{Re}^{+5}(5d^2)$, $\text{W}^{+5}(5d^1)$
- $U(\text{Fe}, \text{Cr}) = 4,3 \text{ eV}$, $J(\text{Fe}, \text{Cr}) = 0.89, 0.87 \text{ eV}$

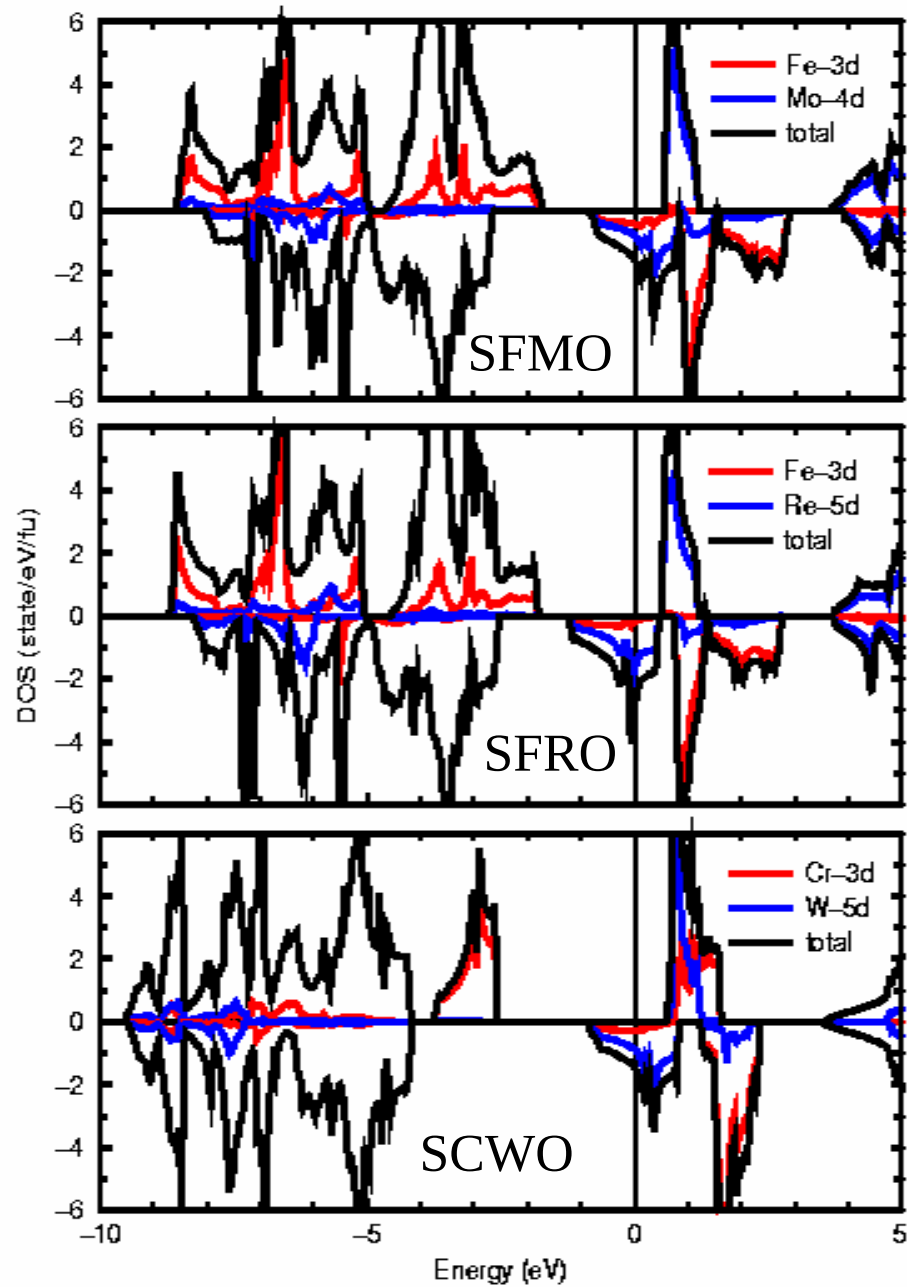
Double perovskite structure



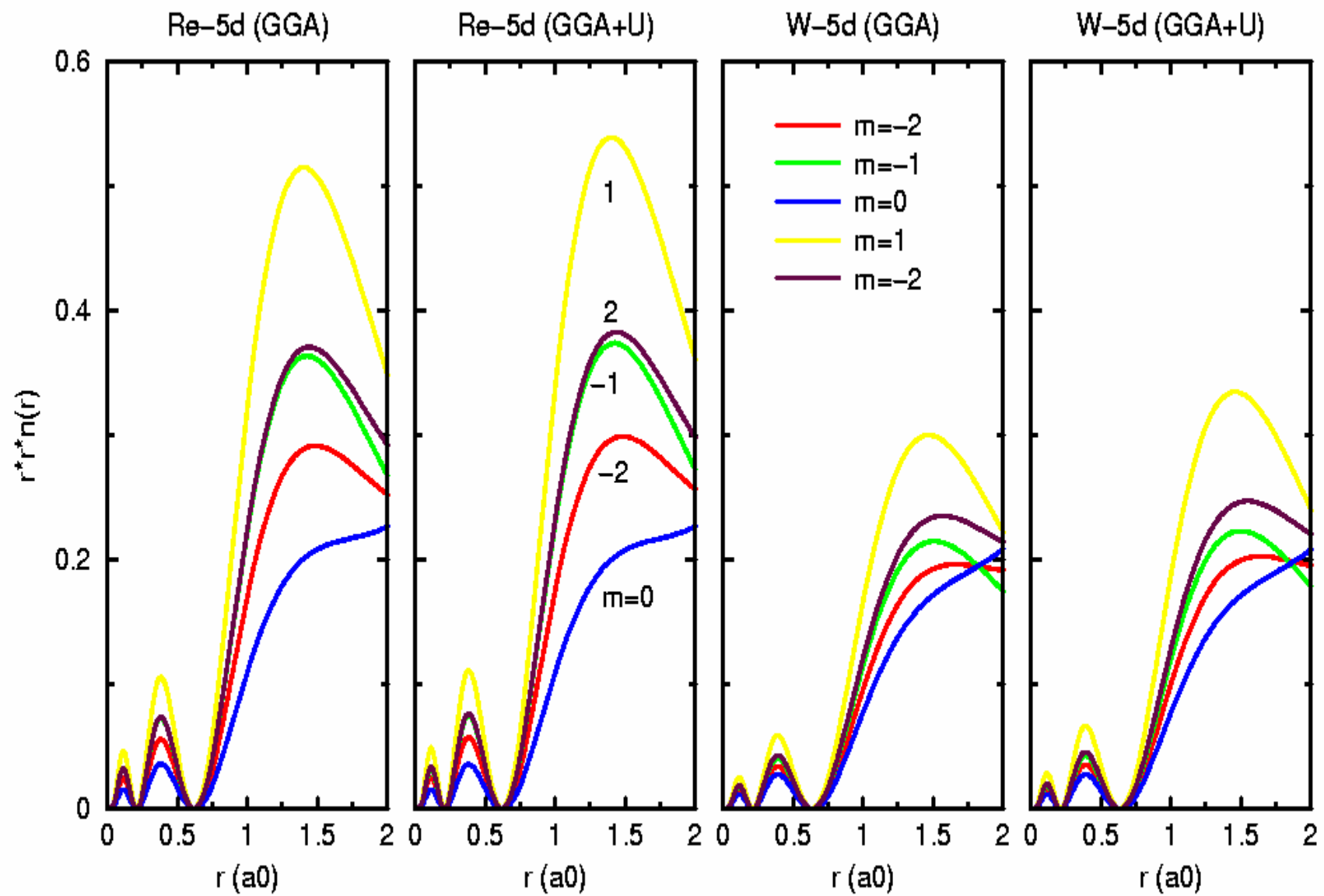
LDA



LDA+U



	spin moment			orbital moment	
Sr ₂ FeMoO ₆ GGA GGA+U	Fe	Mo	total	Fe	Mo
	3.80	-0.33	4.00	0.043	0.032
	3.96	-0.43	4.00	0.047	0.045
Sr ₂ FeReO ₆ GGA GGA+U	Fe	Re	total	Fe	Re
	3.81	-0.85	3.00	0.070	0.23
	3.98	-0.96	3.00	0.066	0.27
Sr ₂ CrWO ₆ GGA GGA+U	Cr	W	total	Cr	W
	2.30	-0.33	2.00	-0.007	0.10
	2.46	-0.45	2.00	-0.007	0.15



Jeng et al PRB (2003)

Double perovskites :

$\text{Sr}_2\text{FeMoO}_6$, $\text{Sr}_2\text{FeReO}_6$, Sr_2CrWO_6

- LDA+U has significant effects on DOS, but experimental data is not available
- Orbital moment of 3d and 4d elements are all quenched because of strong crystal field
- 5d elements exhibit large unquenched orbital moment because of strong spin-orbit interaction in 5d orbitals

SrRuO_3

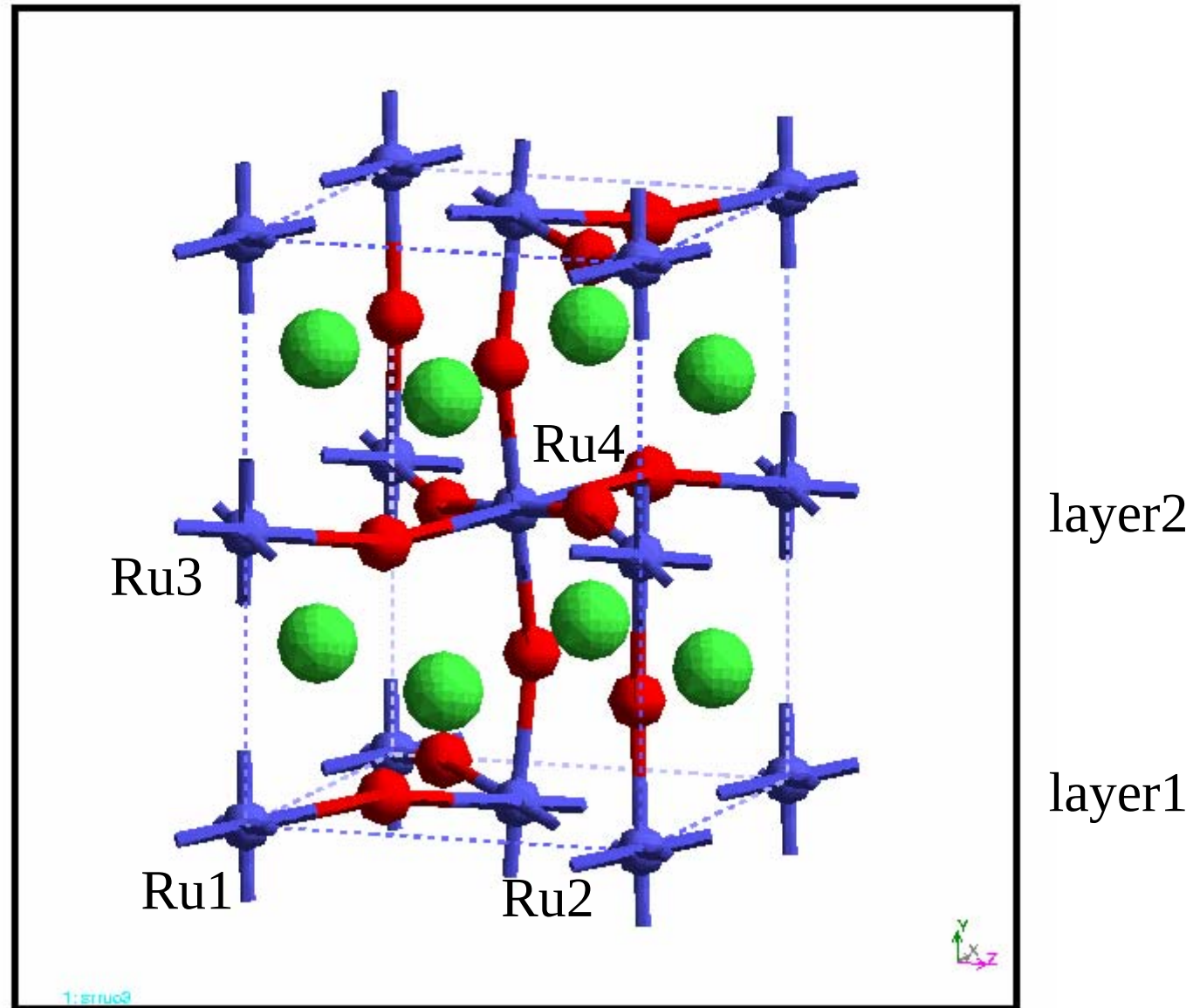
- The only ferromagnetic conductor of 4d oxides
- Lattice type : orthorhombic perovskite
- Space group : Pnma (No. 62)
- 20 atoms in orthorhombic unit cell
- $a=5.5332$ Å, $b=5.57169$ Å, $c=7.8491$ Å
- Ionic model : $\text{Ru}^{4+}(4d^4, t_{2g}^3 \uparrow, t_{2g}^1 \downarrow)$
- $U = 3.5$ eV, $J = 0.58$ eV
- Pseudopotential, 100 k-point, 31360 plane wave, Cut-off energy = 400 eV

Orthorhombic perovskite SrRuO_3

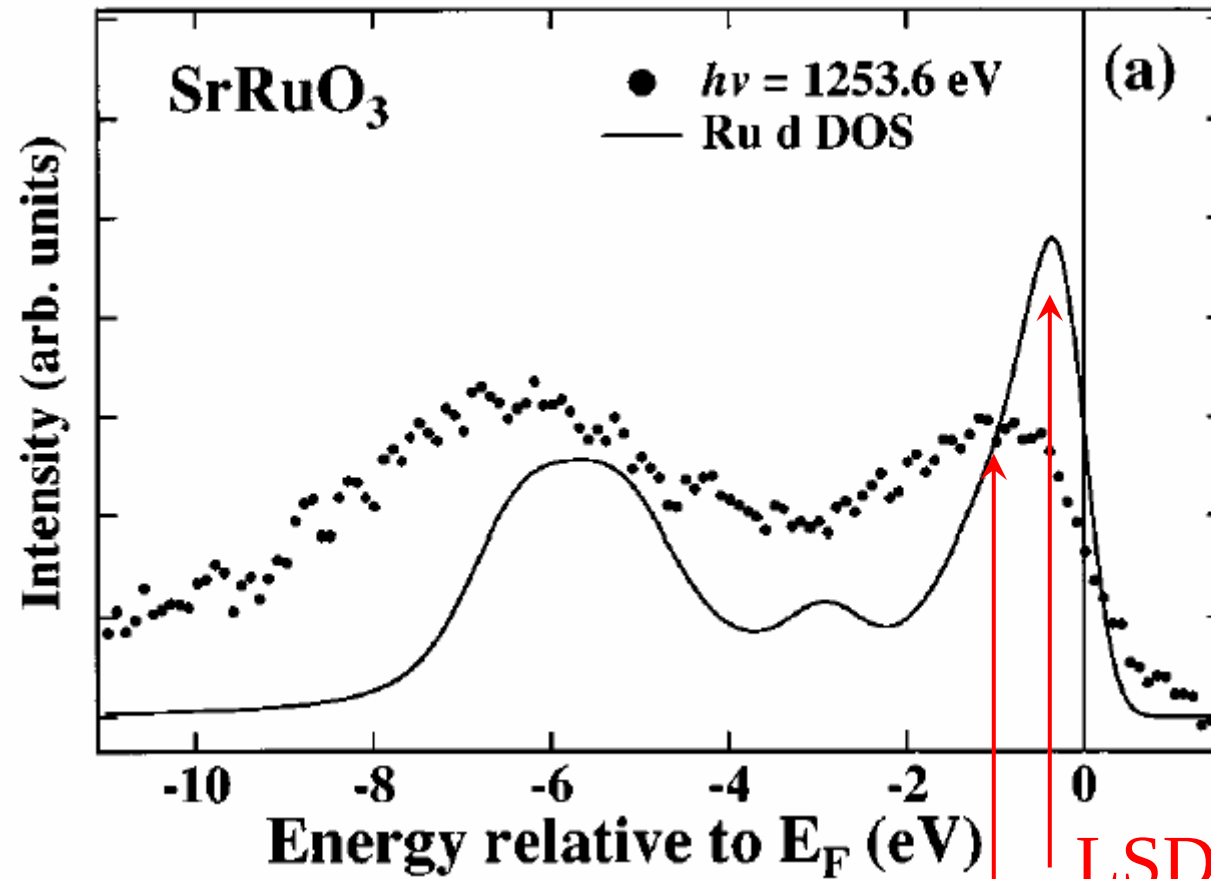
Sr

Ru

O

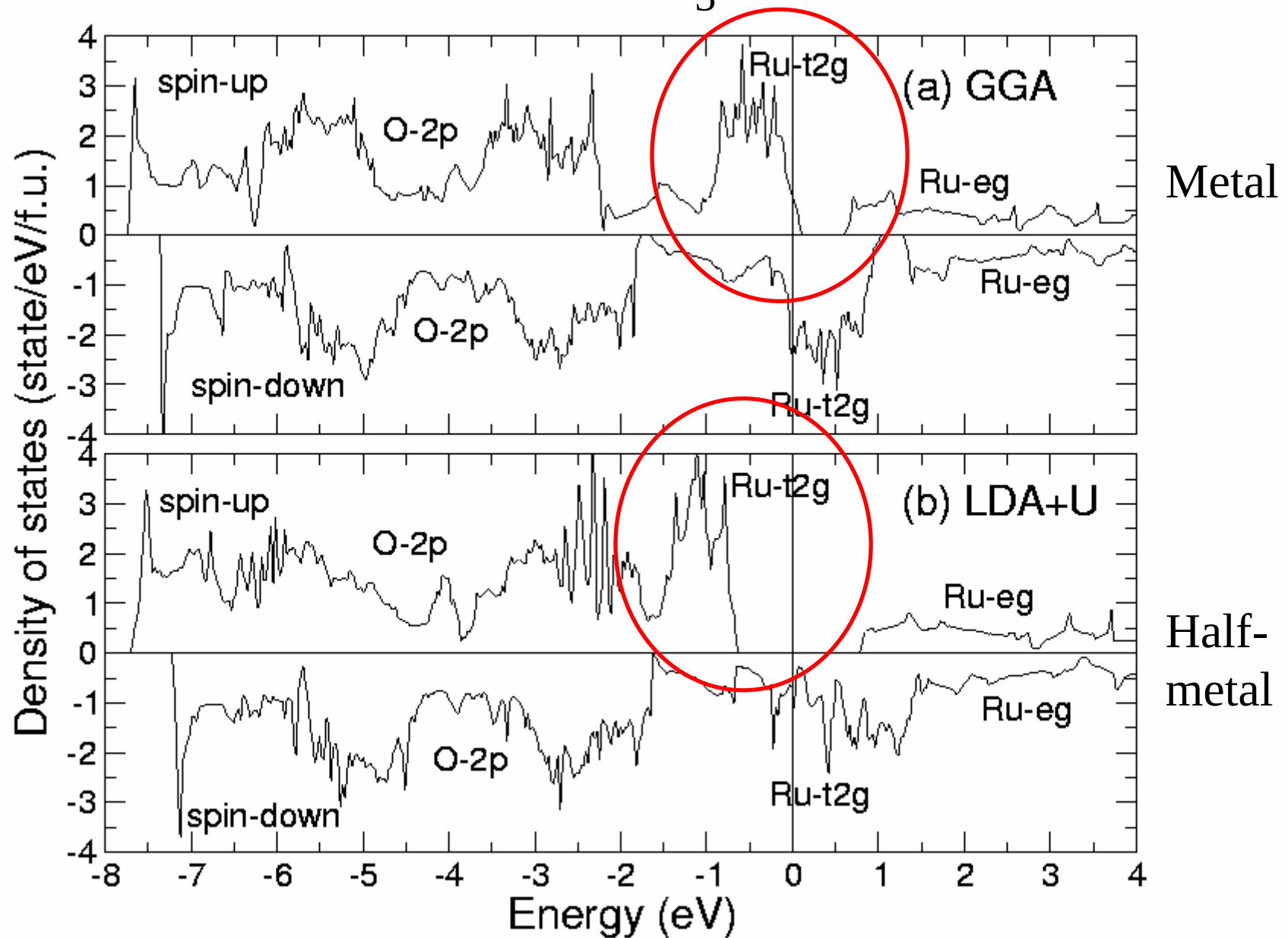


PRB56(1997)6380

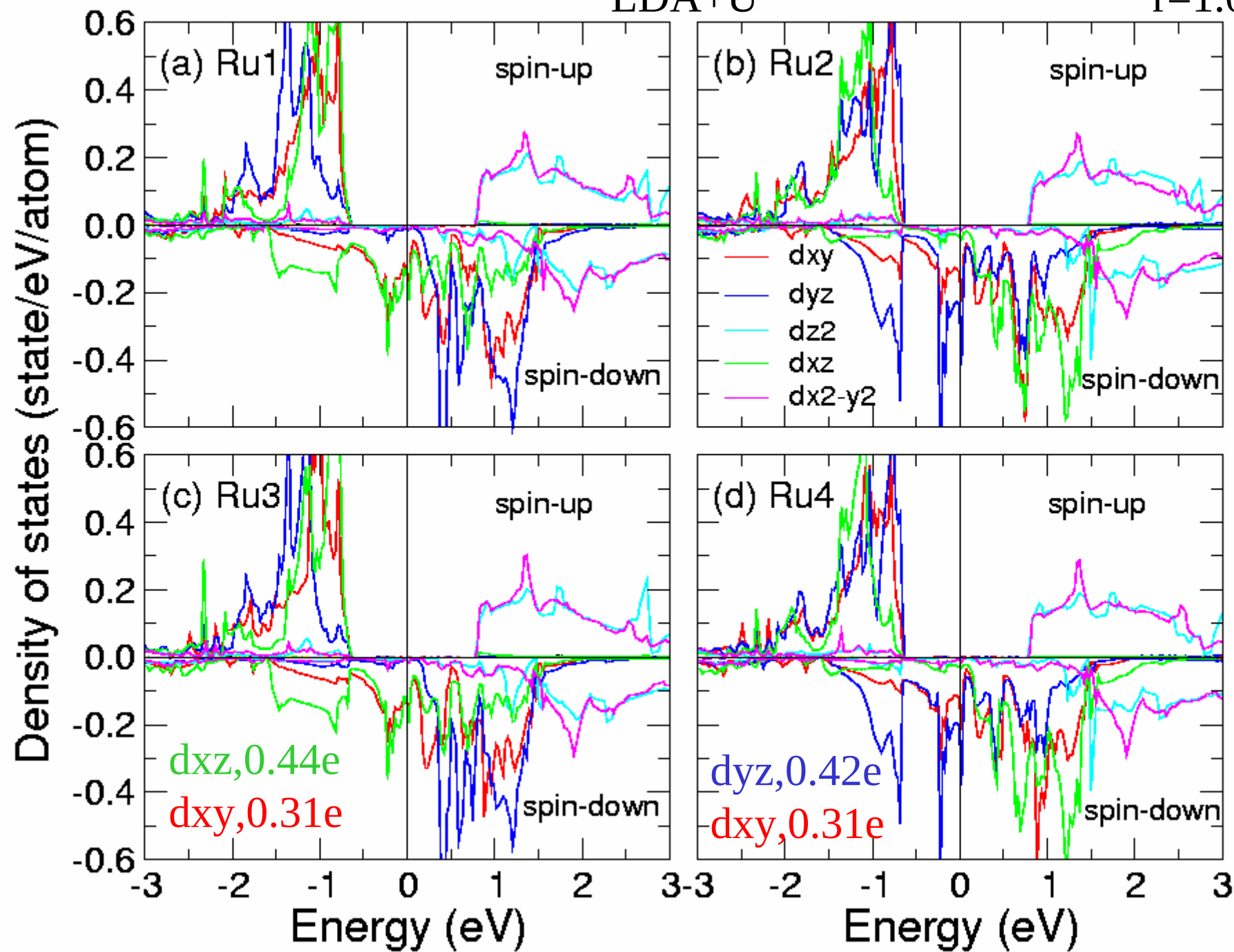


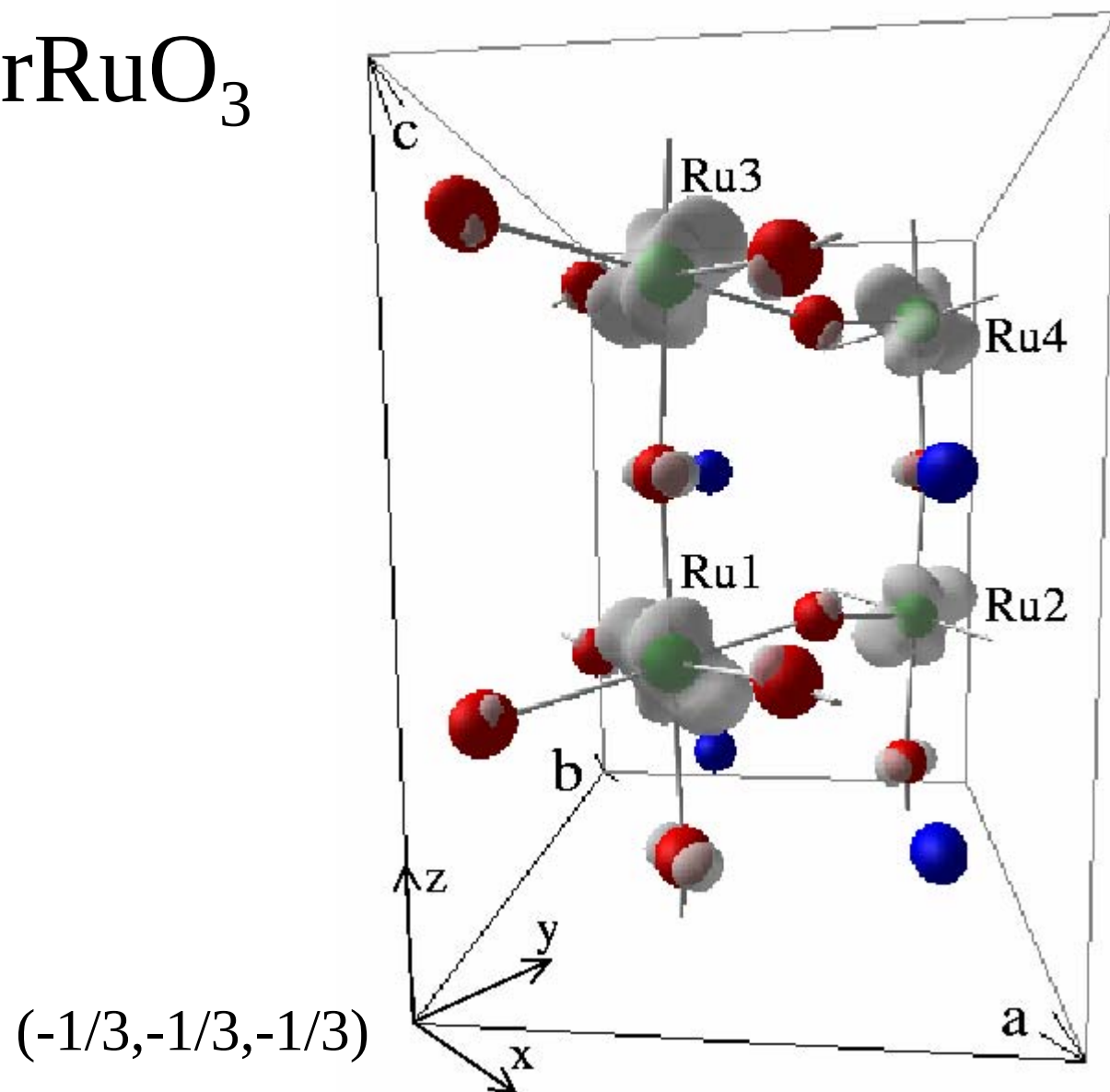
LSDA ~ -0.5 eV

Expt. ~ -1.1 eV



LDA+U

 $r=1.0\text{\AA}$ 

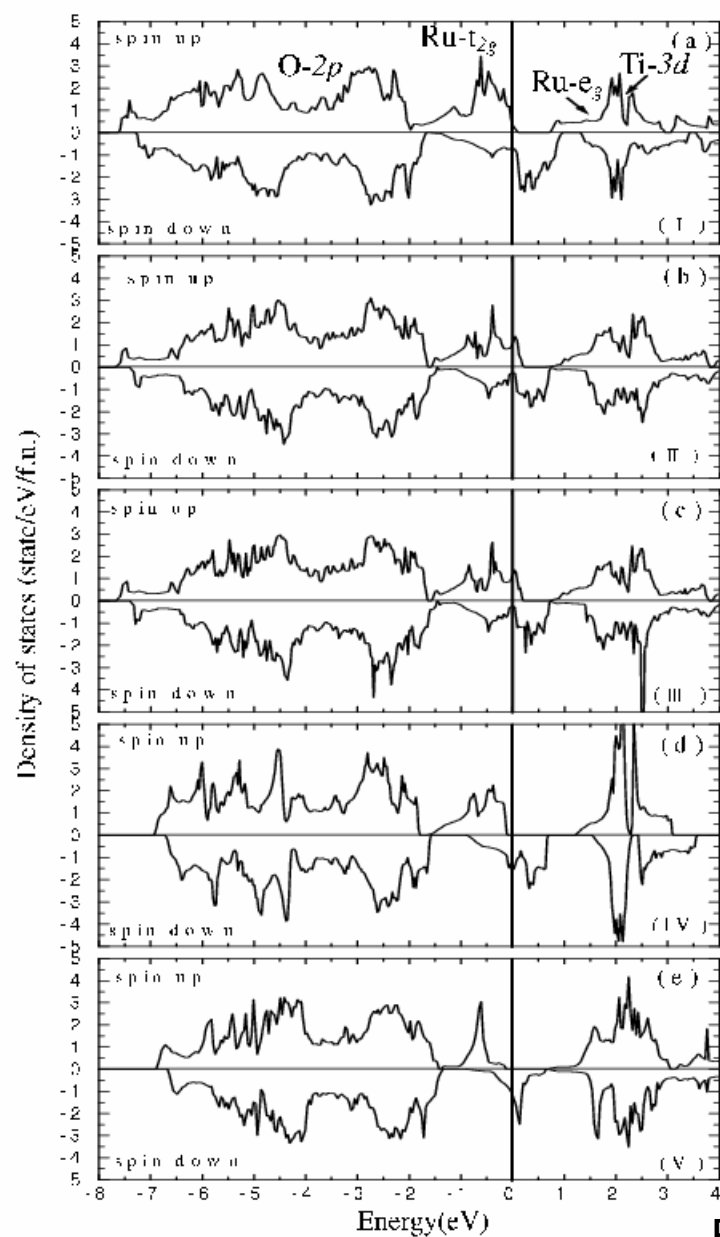
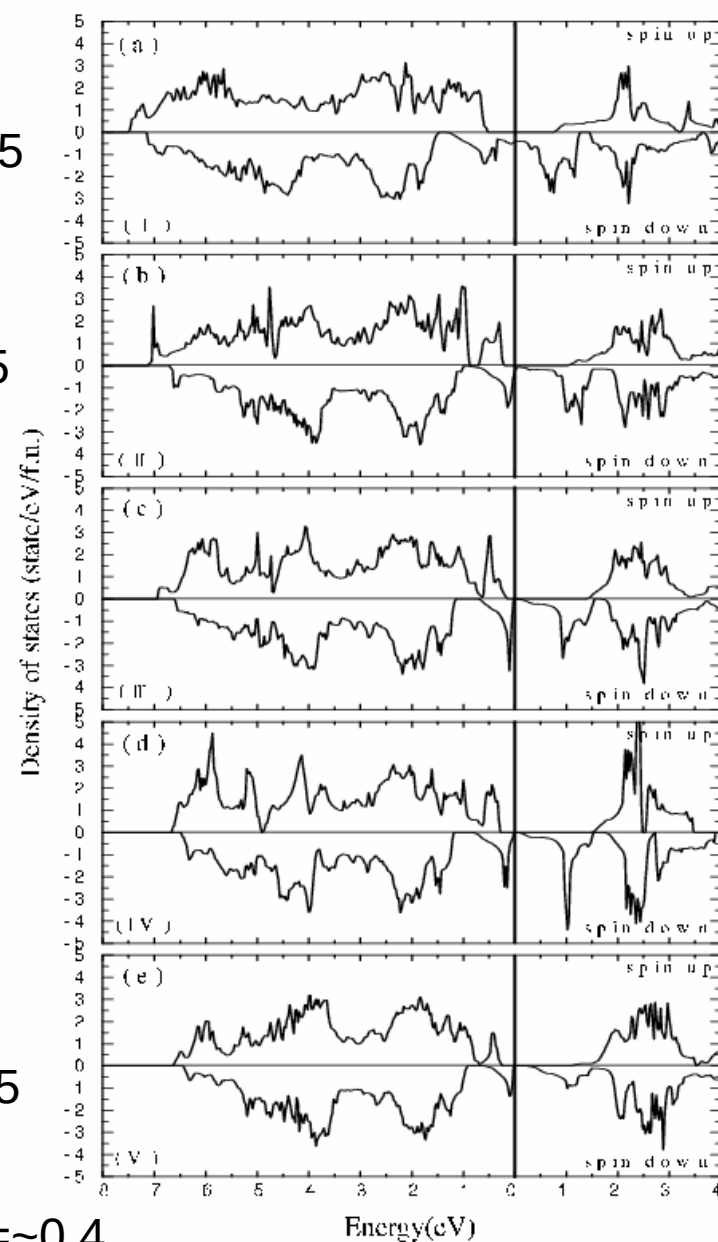


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PRL97(200
6)67002

GGA

MIT in $\text{SrRu}_{1-x}\text{Ti}_x\text{O}_3$

GGA+U

 $x=0.25$ $x=0.5$ $x=0.75$ Exp. MIT $x \sim 0.4$ 



- GGA $\rightarrow$ SrRuO_3 is a normal metal without orbital ordering
- LDA+U $\rightarrow$ SrRuO_3 is a half-metal with $\downarrow$ 4d-t_{2g} orbital ordering
- LDA+U successfully gives the metal-insulator transition upon doping Ti in SrRuO_3 , indicating the strong correlation is important in this system

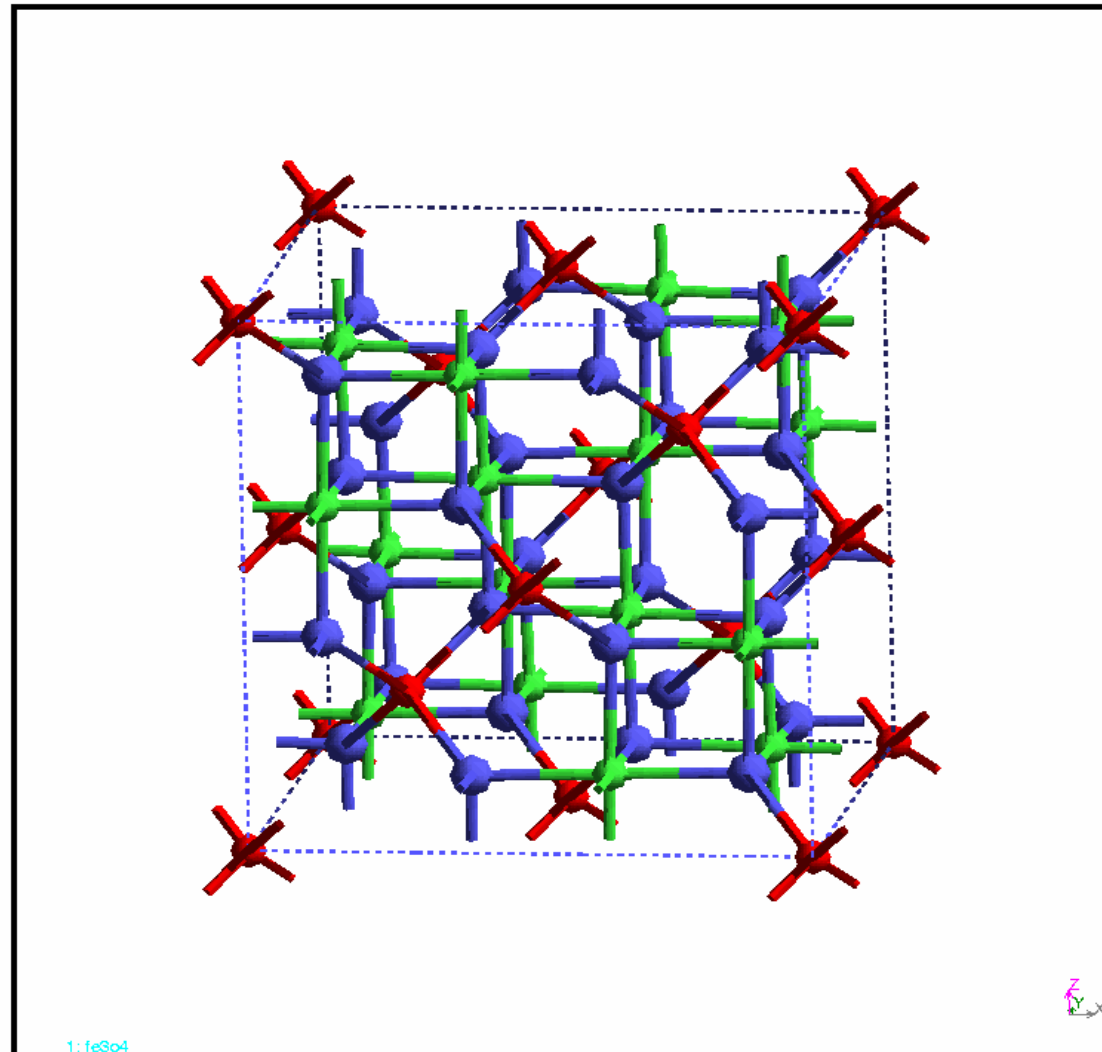
High temperature magnetite (Fe₃O₄)

- Half-metal, insulator, moment = $4 \mu_B$
- Lattice type : fcc
- Space group : Fd3m
- 56 atoms in fcc unit cell
- $a = 8.394, 8.383, 8.351 \text{ \AA}$
- Ionic model : Fe⁺³(3d⁵), Fe⁺²(3d⁶)
- $U(\text{Fe}^{+3}, \text{Fe}^{+2}) = 4.5, 4.0 \text{ eV}$
- $J(\text{Fe}) = 0.89 \text{ eV}$

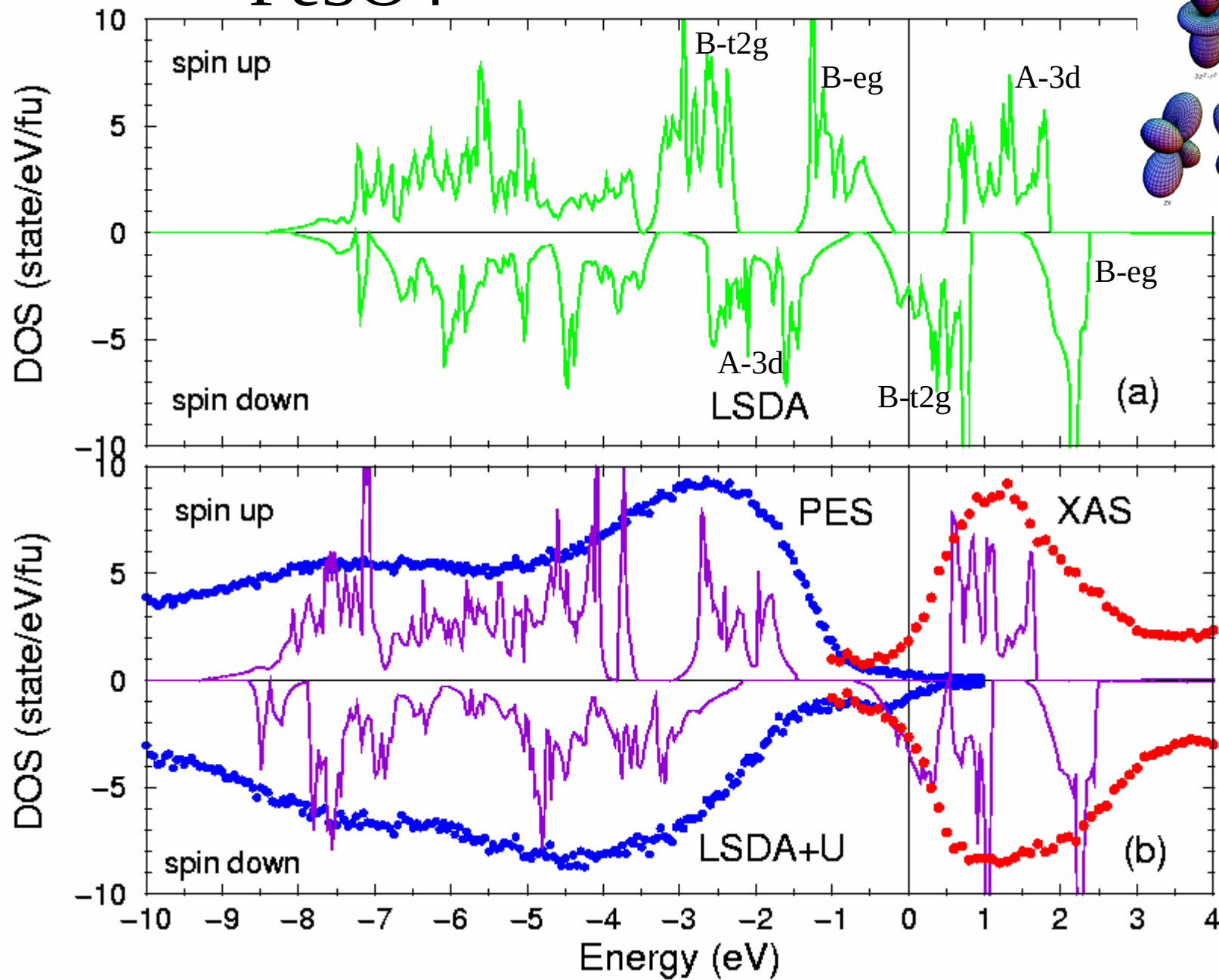
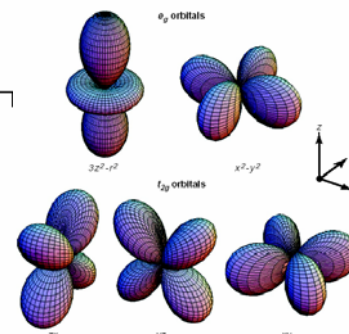
Spinel structure

Author:

Date: Sun May 21 16:31:45 2000



Fe₃O₄ (PES, XAS: D.J. Huang, SRRC)



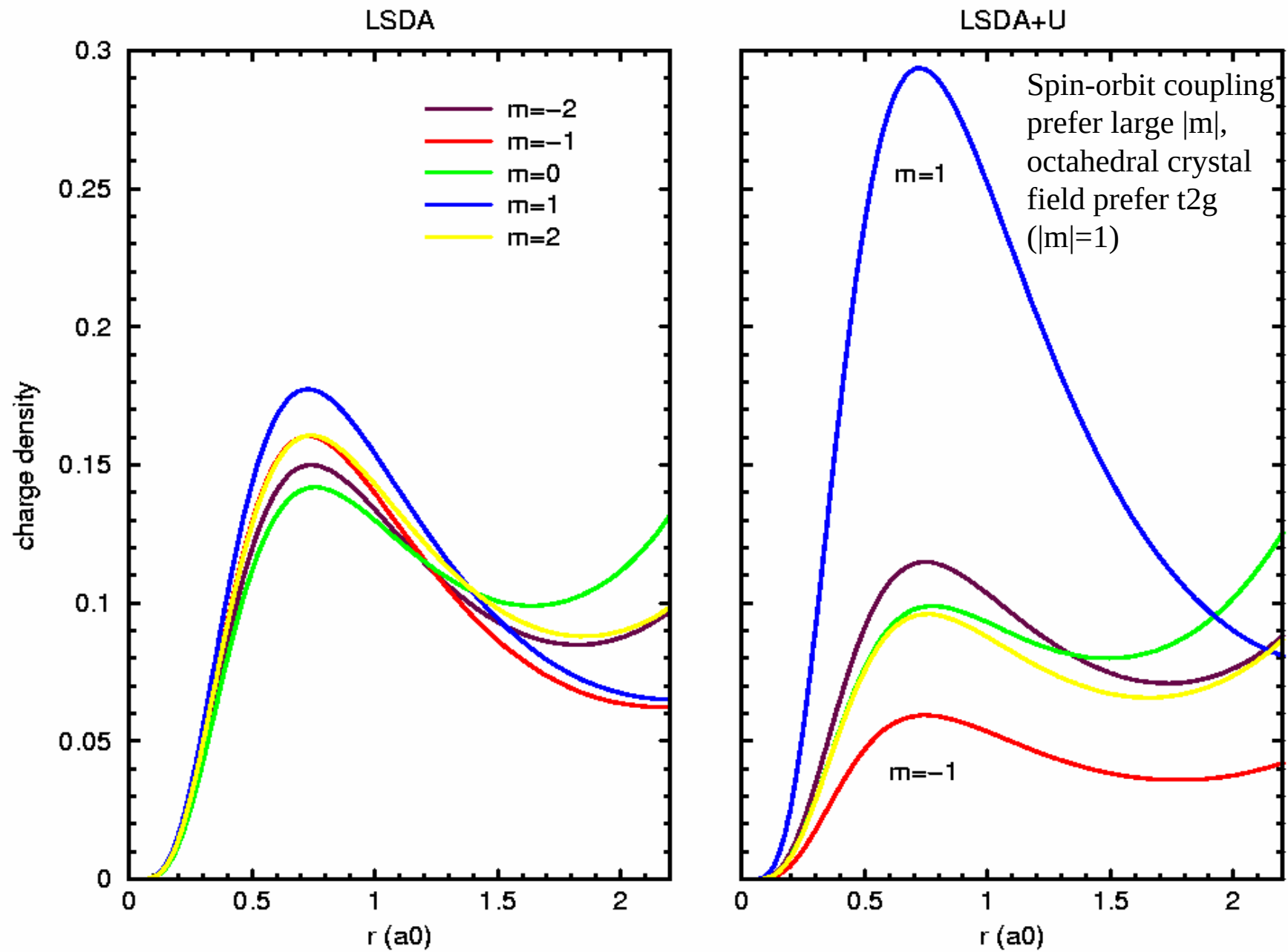
Spin and orbital magnetic moments of Fe₃O₄

(u _B)	Spin moment			Orbital moment	
	Fe(A)	Fe(B)	Total	Fe(A)	Fe(B)
LDA	-3.31	3.52	4.00	-0.018	0.039
GGA	-3.44	3.60	4.00	-0.016	0.038
LDA+U	-3.81	3.81	4.00	-0.014	0.21
Exp.	-3.8 ⁺		4.1		0.2~0.4 [*]

⁺J. Phys. C 11 (1978) 4389

^{*}D. J. Huang et al, SRRC

Fe_3O_4 , PRL93(2004)77204



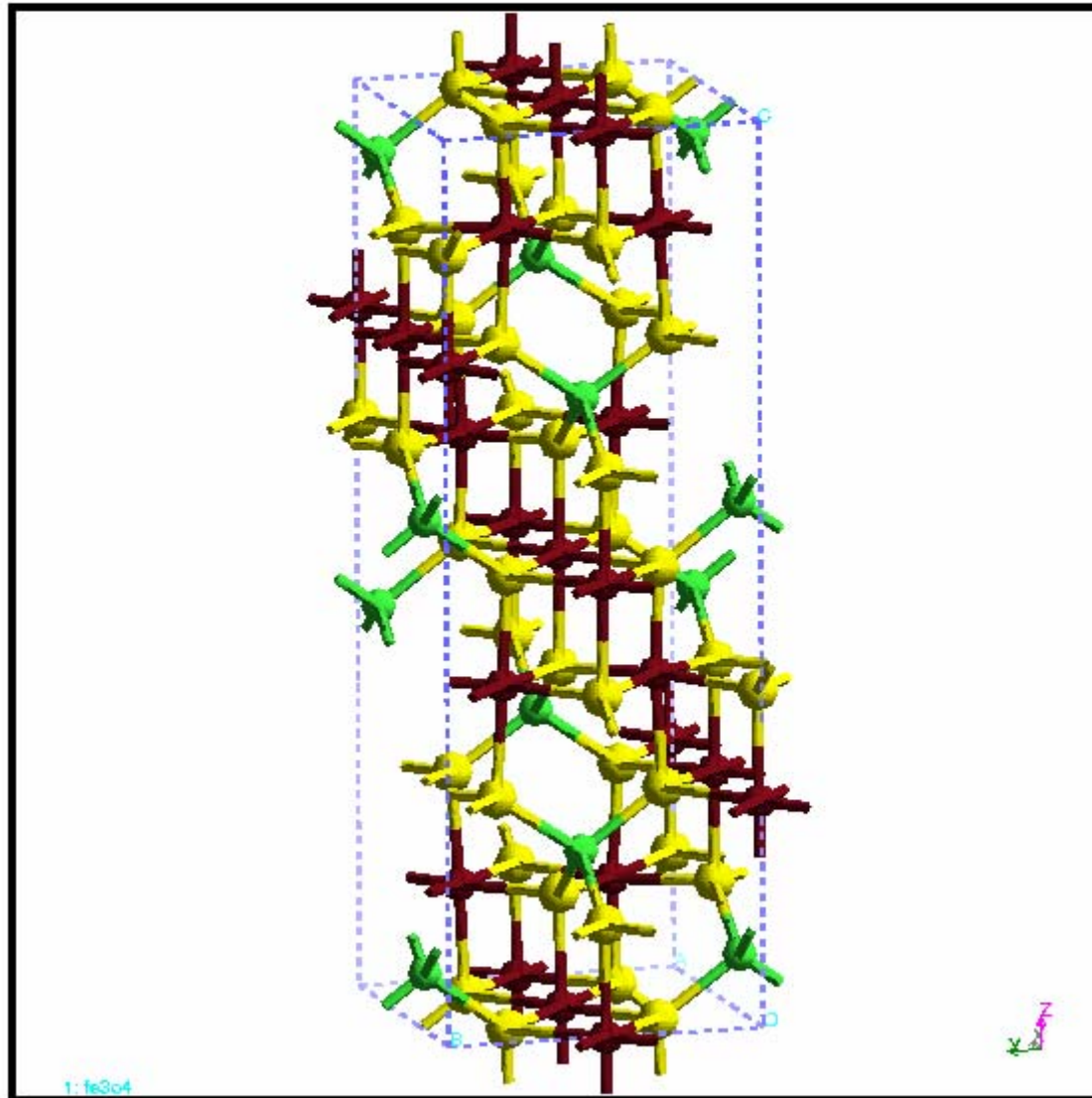
High temperature magnetite (Fe_3O_4)

- LDA+U gives better DOS for Fe_3O_4
- Spin moment of Fe_3O_4 from LDA+U agrees better with neutron scattering measurement
- On-site U gives large unquenched orbital magnetic moments for Fe(B) compatible with MCD results

Low-temperature Fe_3O_4

- Insulator
- Lattice type : monoclinic
- Space group : P2/c
- 56 atoms in monoclinic unit cell
- $a = 5.9444 \text{ \AA}$, $b = 5.925 \text{ \AA}$, $c = 16.775 \text{ \AA}$
- $\beta = 90.2363^\circ$
- Ionic model : $\text{Fe}^{+3}(3d^5)$, $\text{Fe}^{+2}(3d^6)$
- $U = 4.5 \text{ eV}$, $J = 0.89 \text{ eV}$
- Pseudopotential (75600 plane wave up to 400eV)

P2/C structure (low-temperature magnetite)



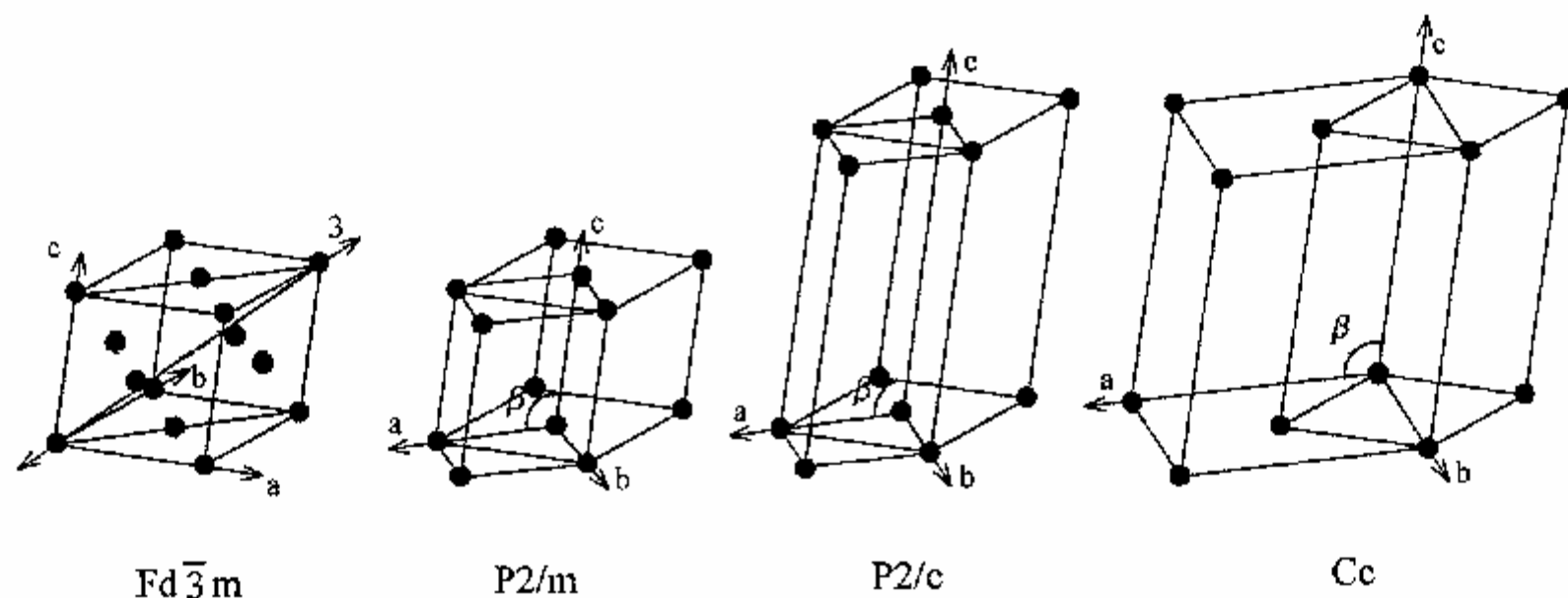


FIG. 2. Relationships between the unit cells referred to in the text.

grown by the skull melter technique. A trace of Fe_2O_3 (0.8 wt % by Rietveld analysis), perhaps arising from surface oxidation of the powder during grinding, was found in the diffraction patterns for this sample. High-resolution neutron powder-diffraction data were collected at 90 and 130 K using the high-resolution neutron powder-diffractometer (HRPD) instrument at the ISIS spallation source, United Kingdom.³³ Approximately 10 g of sample was loaded into a vanadium sample can and mounted in a closed-cycle refrigerator to maintain temperature control while retaining a low instrumental background. Data were collected at 90 K for a total of

further distortions were required to account for the peak broadenings and splittings. Figure 2 shows the relationships between the unit cells in the low- and high-temperature phases.

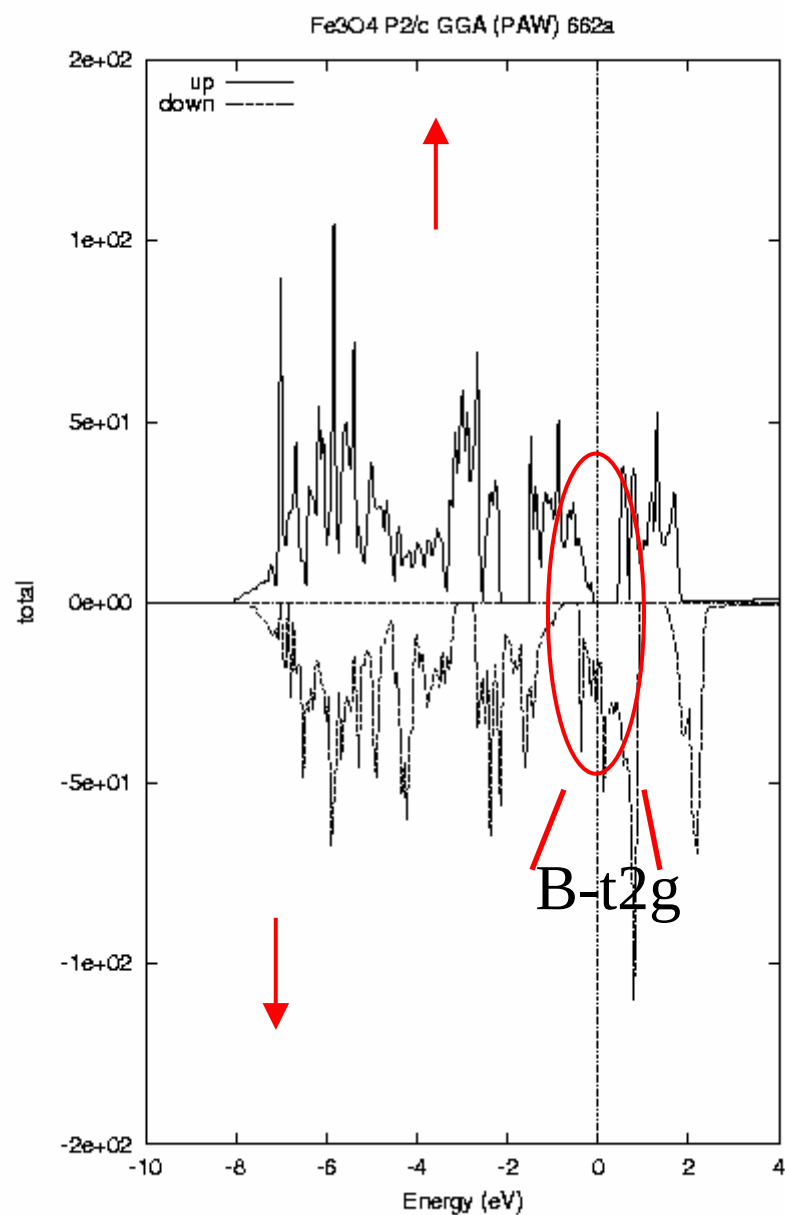
Numerous superstructure peaks were identified by comparing the 90 and 130 K profiles. All of these superstructure peaks were indexed on the monoclinic unit cell previously proposed,⁸ which has Cc symmetry and axes;

$$a_m = -(a_c + b_c),$$

$$b_m = (a_c - b_c),$$

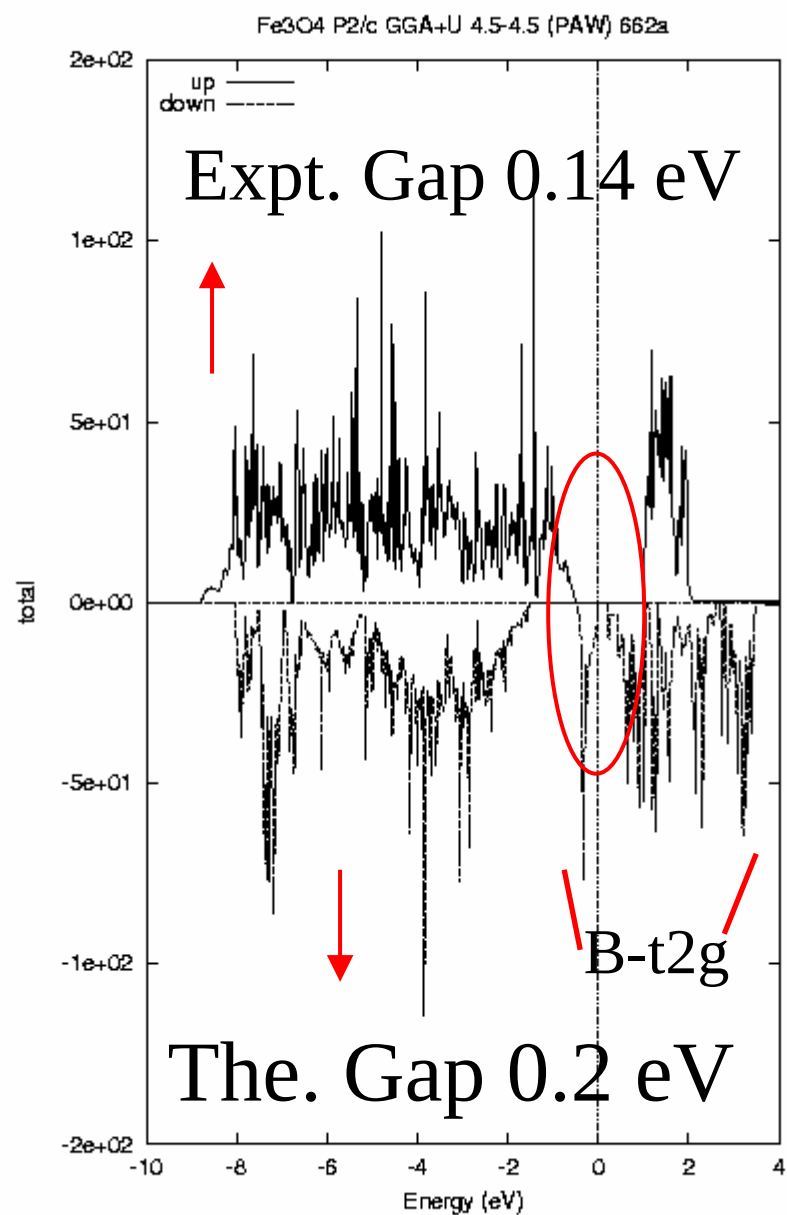
LDA

Half-metal



LDA+U

Insulator

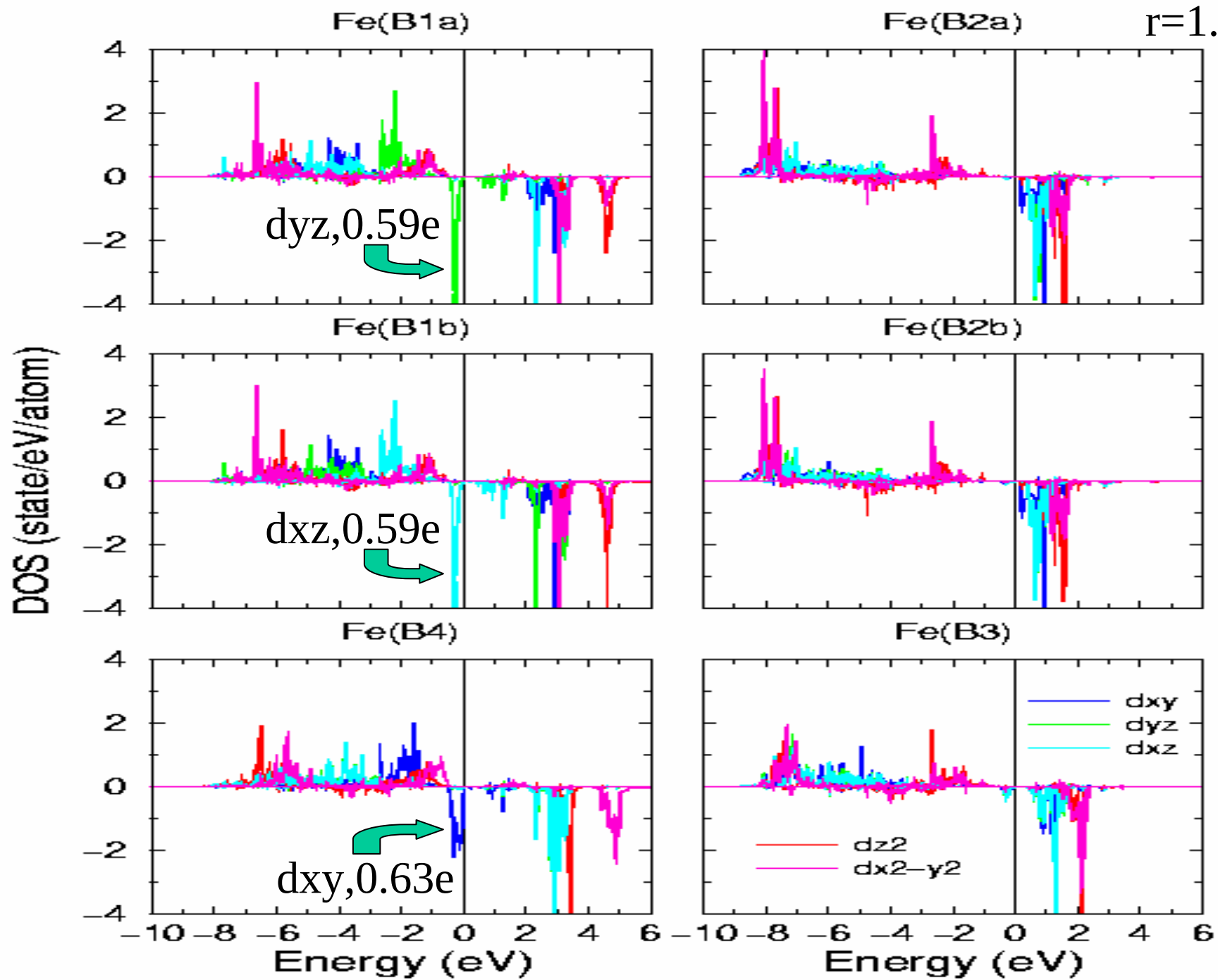


Charge and spin moment of Fe_3O_4
in low-temperature phase ($r_{\text{mt}} = 1.0 \text{ \AA}$)

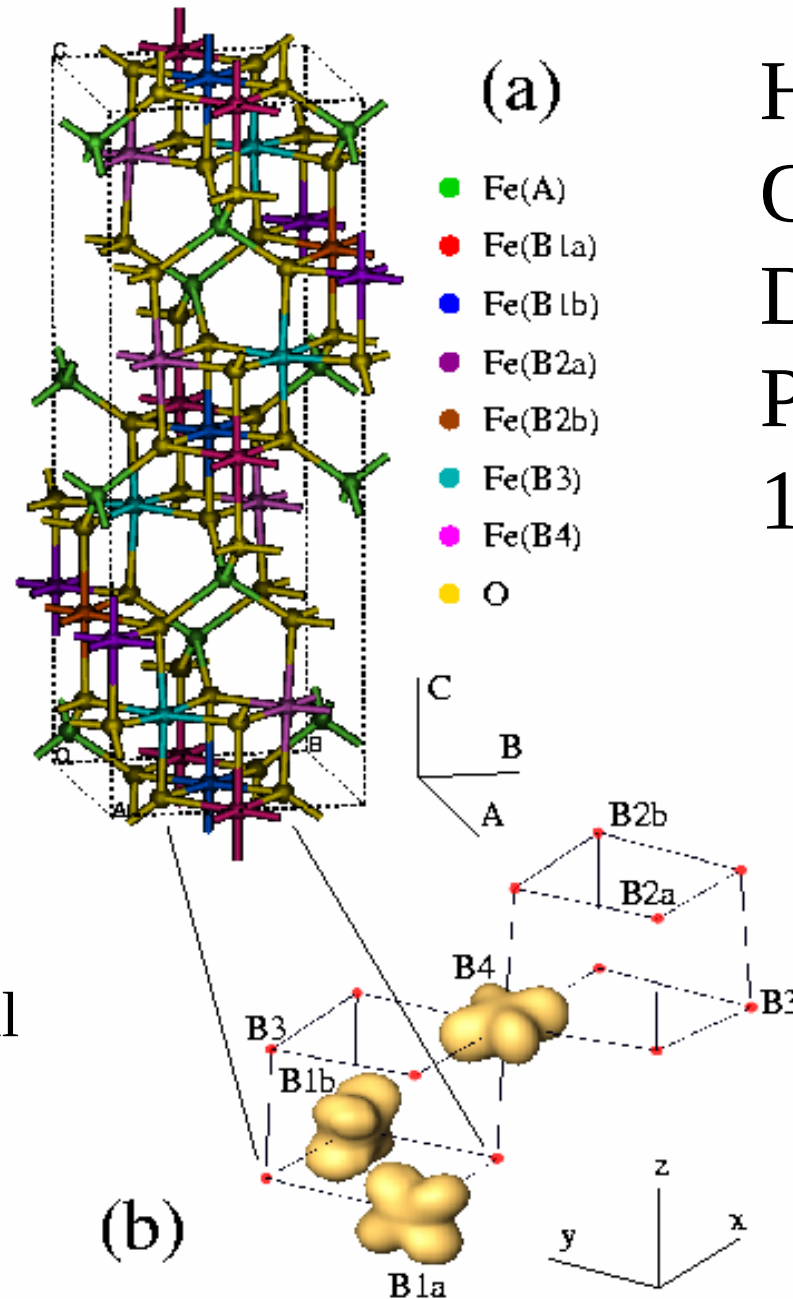
	LDA	LDA	Exp.*	LDA+ U	LDA+ U
	Spin	Charge	Charge	Charge	Spin
Fe(B1)	3.34	5.53	5.6	5.57	3.45
Fe(B2)	3.43	5.50	5.4	5.41	3.90
Fe(B3)	3.32	5.52	5.4	5.44	3.81
Fe(B4)	3.39	5.52	5.6	5.58	3.39

* PRL 87 (2001) 266401

$r=1.0\text{\AA}$



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G. Y. Guo,
D. J. Huang,
PRL93(2004)
156403,



Low-temperature Fe₃O₄

- LSDA + lattice distortion → half-metal, no charge ordering
- LSDA+U + lattice distortion → insulating ground state with charge-orbital ordering

Conclusions

- Transition-metal oxides
 - narrower bandwidth
 - closer to localized electron picture
 - more or less strongly correlated
 - LDA+U gives better result
 - the on-site Coulomb correlation U is important in these systems

CMS II-6 Hands-on: LDA+U, orbital decomposed DOS, and band decomposed charge distribution

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Working list

- Setup LDA+U bulk calculations for NiO(AFMII)
- Calculate orbital decomposed DOS for NiO(AFMII)
- Calculate band decomposed charge and spin density distribution for NiO(AFMII)

Setup LDA+U bulk calculations for NiO(AFMII)

INCAR:

SYSTEM = NiO FCC

DOS related values

LDAU = .TRUE.

LDAUTYPE = 1

LDAUL = 2 -1

LDAUU = 8.0 0.0

LDAUJ = 0.95 0.0

LMAXMIX = 4

LDAUPRINT = 2

ISMear = -5

ISPIN = 2

MAGMOM = 2 -2 2*0

GGA = 91

RWIGS = 1.2 0.8

KPOINTS, POSCAR, and
POTCAR are the same

Switch on L(S)DA+U calculation

Setup rotationally invariant L(S)DA+U scheme

+U in d(l=2) orbital, no on-site U if -1

U=8 and 0eV for Ni and O, respectively

J=0.95 and 0eV for Ni and O, respectively

Expand charge in Ylm up to l=4 (for d orbital)

Output occupation matrix

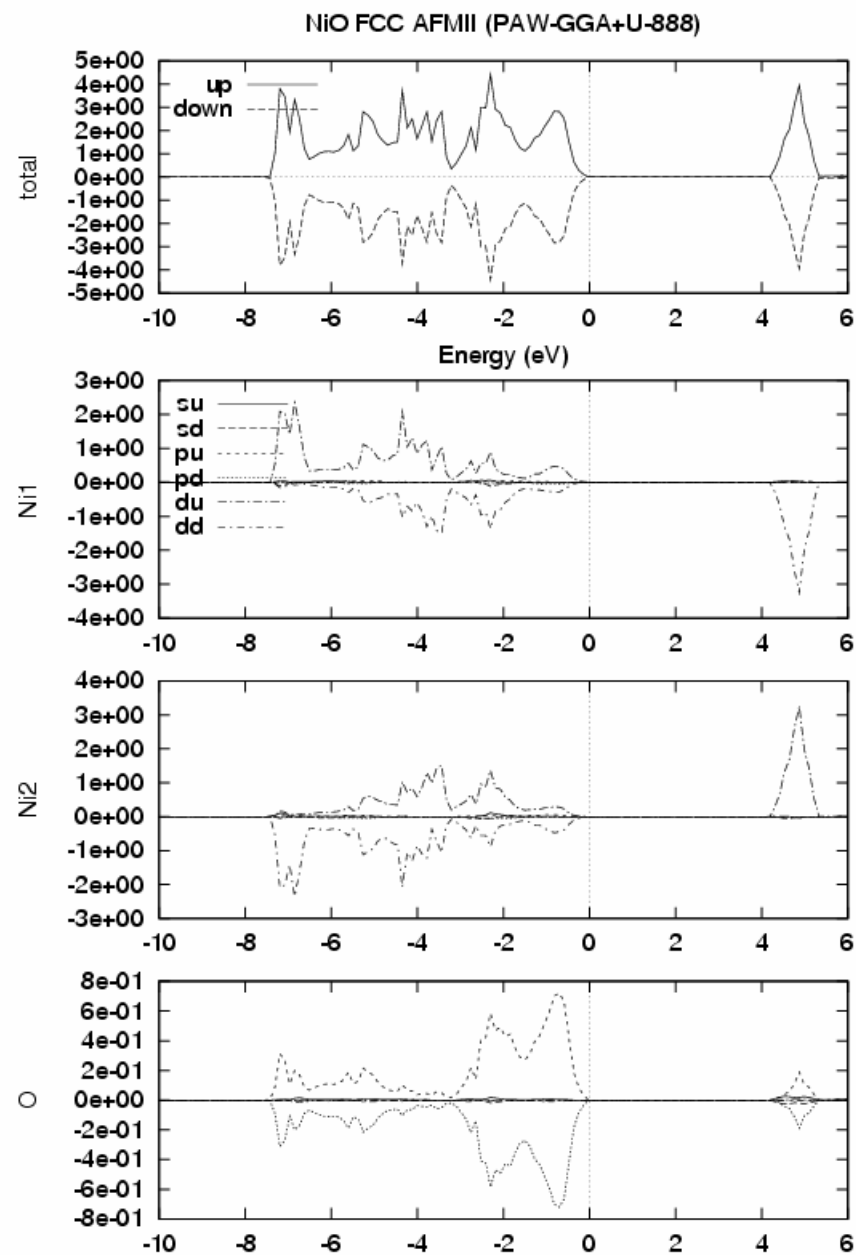
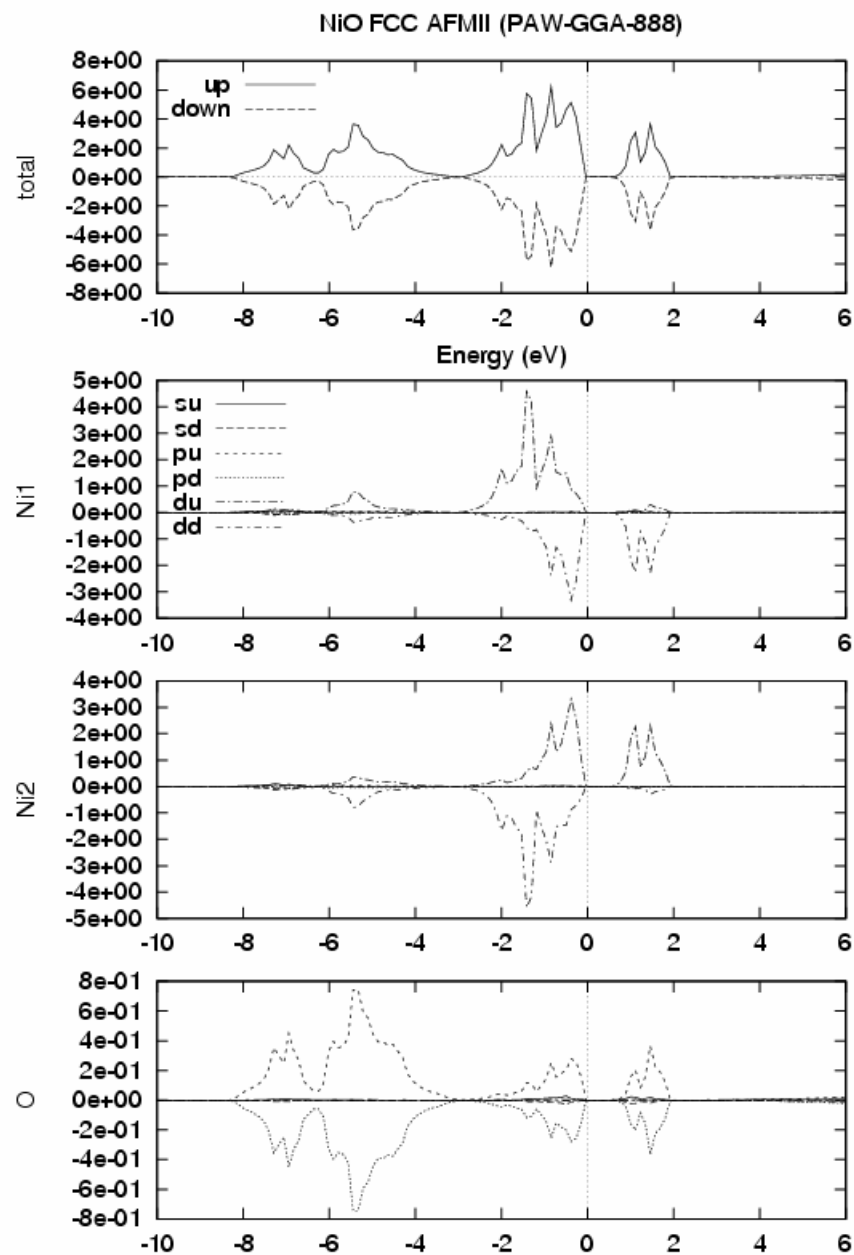
*U and J must be specified for all atomic types

LDA+U controlling parameters

The L(S)DA+U in VASP is switched on by means of the following tags

- LDAU = .TRUE. Switches on the L(S)DA+U.
- LDAUTYPE = 1|2|4 Type of L(S)DA+U (Default: LDAUTYPE = 2)
 - 1 Rotationally invariant LSDA+U according to Liechtenstein *et al.*
 - 4 Idem 1., but LDA+U instead of LSDA+U (i.e. no LSDA exchange splitting)
 - 2 Dudarev's approach to LSDA+U (Default)
- LDAUL = L ... *l*-quantum number for which the on site interaction is added
(-1: no on site terms added, 1: p, 2: d, 3: f, Default: LDAUL = 2)
- LDAUU = U ... Effective on site Coulomb interaction parameter
- LDAUJ = J ... Effective on site Exchange interaction parameter
- LDAUPRINT = 0|1|2 Controls verbosity of the L(S)DA+U module
(0: silent, 1: Write occupancy matrix to OUTCAR, 2: idem 1., plus potential matrix dumped to stdout,
Default: LDAUPRINT = 0)

NB: LDAUL, LDAUU, and LDAUJ must be specified for *all* atomic species!



Orbital decomposed DOS for NiO(AFMII)

INCAR:

SYSTEM = NiO FCC

DOS related values

LORBIT = 1

LDAU = .TRUE.

LDAUTYPE = 1

LDAUL = 2 -1

LDAUU = 8.0 0.0

LDAUJ = 0.95 0.0

LMAXMIX = 4

LDAUPRINT = 2

ISMear = -5

ISPIN = 2

MAGMOM = 2 -2 2*0

GGA = 91

RWIGS = 1.2 0.8

Switch on orbital decomposed DOS calculation

Total DOS

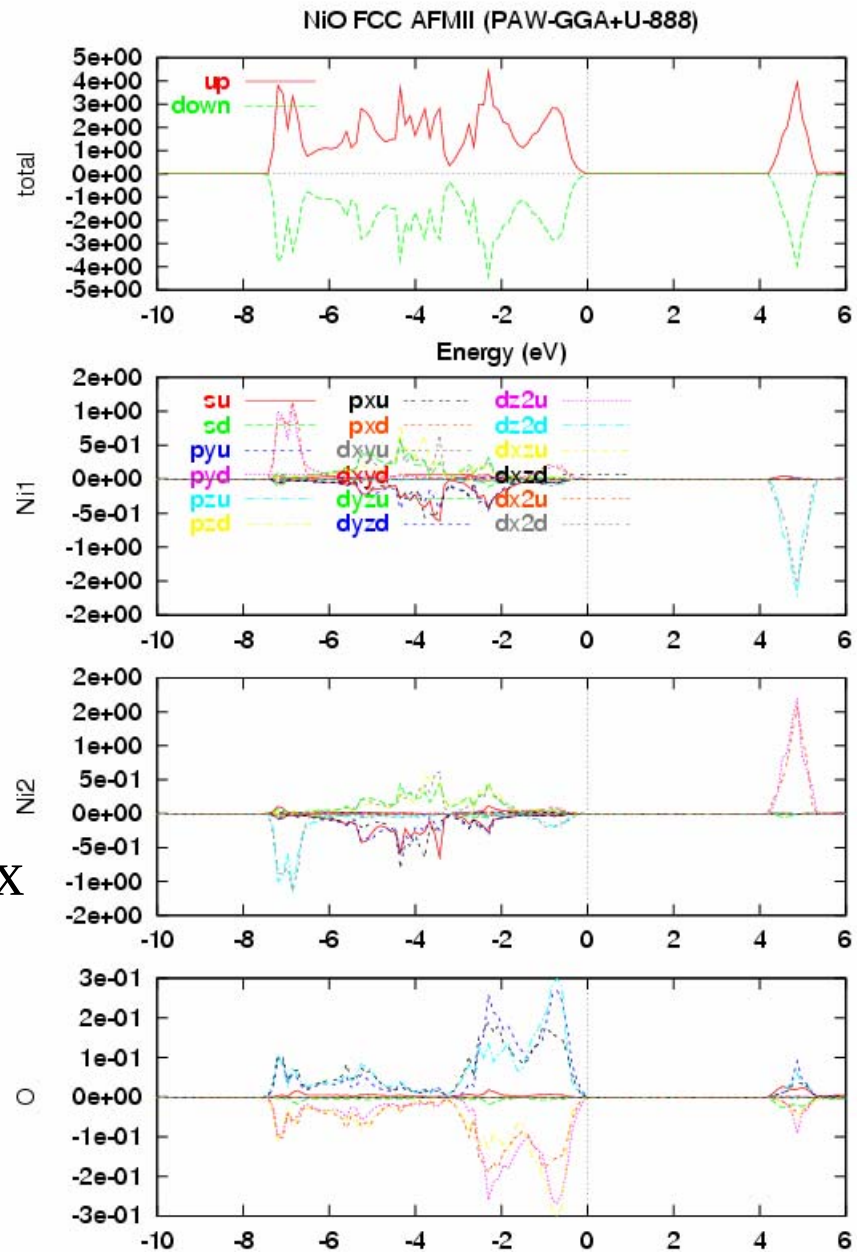
.....								
17.428	0.0000E+00	0.0000E+00	0.2000E+02	0.2000E+02				
17.542	0.0000E+00	0.0000E+00	0.2000E+02	0.2000E+02				
17.656	0.0000E+00	0.0000E+00	0.2000E+02	0.2000E+02				
17.65555450	-16.50226769	301	4.12051746	1.000000000				
-16.502	0.0000E+00	0.0000E+00	0.0000E+00	0.0000E+00	0.0000E+00	0.0000E+00	0.0000E	
-16.388	0.0000E+00	0.0000E+00	0.0000E+00	0.0000E+00	0.0000E+00	0.0000E+00	0.0000E	
-16.275	0.0000E+00	0.0000E+00	0.0000E+00	0.0000E+00	0.0000E+00	0.0000E+00	0.0000E	

PDOS of the 1st atom in order of $((l=2)+1)^2=9$ terms

s	py	pz	px	dxy	dyz	dz2	dxz	dx2
u d	u d	u d	u d	u d	u d	u d	u d	u d

9*(u d)=18 terms

t2g: dxy, dyz, dzx



eg: dz2, dx2-y2

band decomposed charge and spin density distribution for NiO(AFMII)

INCAR:

SYSTEM = NiO FCC

DOS related values

LPARD = .TRUE.

EINT = 4.0 6.0

NBMOD = -3

LDAU = .TRUE.

LDAUTYPE = 1

LDAUL = 2 -1

LDAUU = 8.0 0.0

LDAUJ = 0.95 0.0

LMAXMIX = 4

LDAUPRINT = 2

ISMear = -5

ISPIN = 2

MAGMOM = 5 -5 0 0

GGA = 91

RWIGS = 1.2 0.8

Switch on partial charge distribution calculation

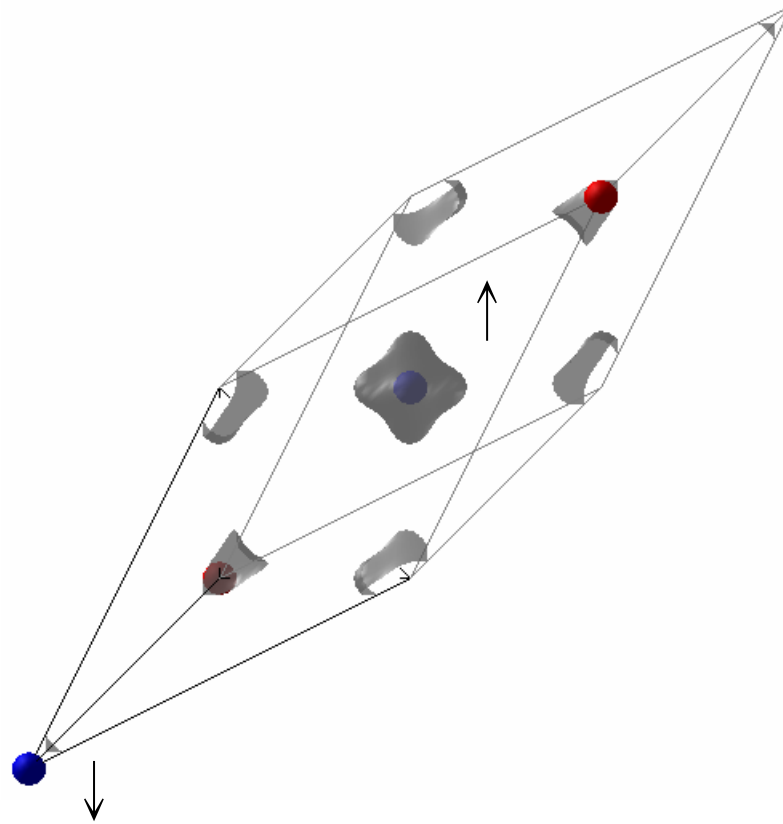
From 4eV to 6eV

With respect to Fermi level

Note: You need to calculate a
Convergent WAVECAR first!!
Then use this WAVECAR to do
this calculation

LDA+U NiO AFMII eg band: 4~6eV charge distribution

vaspview PARCHG



Homework

please Email to jeng@phys.sinica.edu.tw

- Calculate and plot LDA+U DOS of CrO₂, NiO, CoO, and FeO ($U=3, 8, 7.8, 4.5$ eV, $J=0.87, 0.95, 0.92, 0.89$ eV for Cr, Ni, Co, and Fe, respectively)
- Calculate and plot orbital decomposed LDA+U DOS of NiO, CoO, and FeO
- Calculate and plot (vaspview) the charge density distribution of the unoccupied eg band of NiO, CoO, and FeO from LDA+U
- Make a list of the magnetic moment (total and individual) of FeO, CoO, NiO, and CrO₂ from LDA and LDA+U